

# **Nanomagnetism**

## **Part 1: Physics of Nanomagnets**

**Sara A. Majetich**

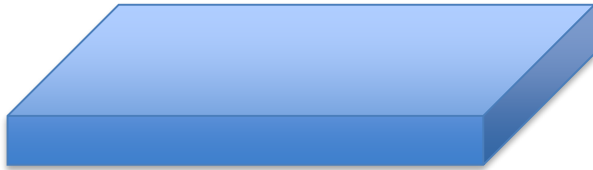
Physics Dept., Carnegie Mellon Univ.

Pittsburgh, PA 15213 USA

sara@cmu.edu

# What Defines Nano?

At least one dimension is  $< 1000$  nm



2 D: Thin film



1 D: nanowire



0 D: nanodot

# Magnetic Length Scales

| Length                              | Symbol                | Typical Magnitude (nm) |
|-------------------------------------|-----------------------|------------------------|
| Exchange length                     | $l_{\text{ex}}$       | 1 - 100                |
| Domain wall width                   | $\delta$              | 1 - 100                |
| Spin diffusion length               | $\lambda_{\text{sd}}$ | 1 - 100                |
| Critical single domain diameter     | $d_{\text{cr}}$       | 10 - 1000              |
| Critical superparamagnetic diameter | $d_{\text{sp}}$       | 1 - 100                |

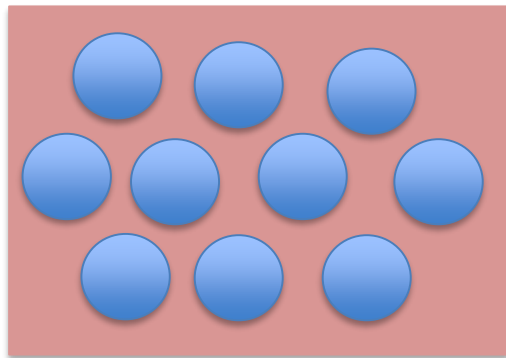
# What's Included Here in Nanomagnetism?



*Nanoparticles*



Nanowires



[Nanocomposites]



Nanopatterns

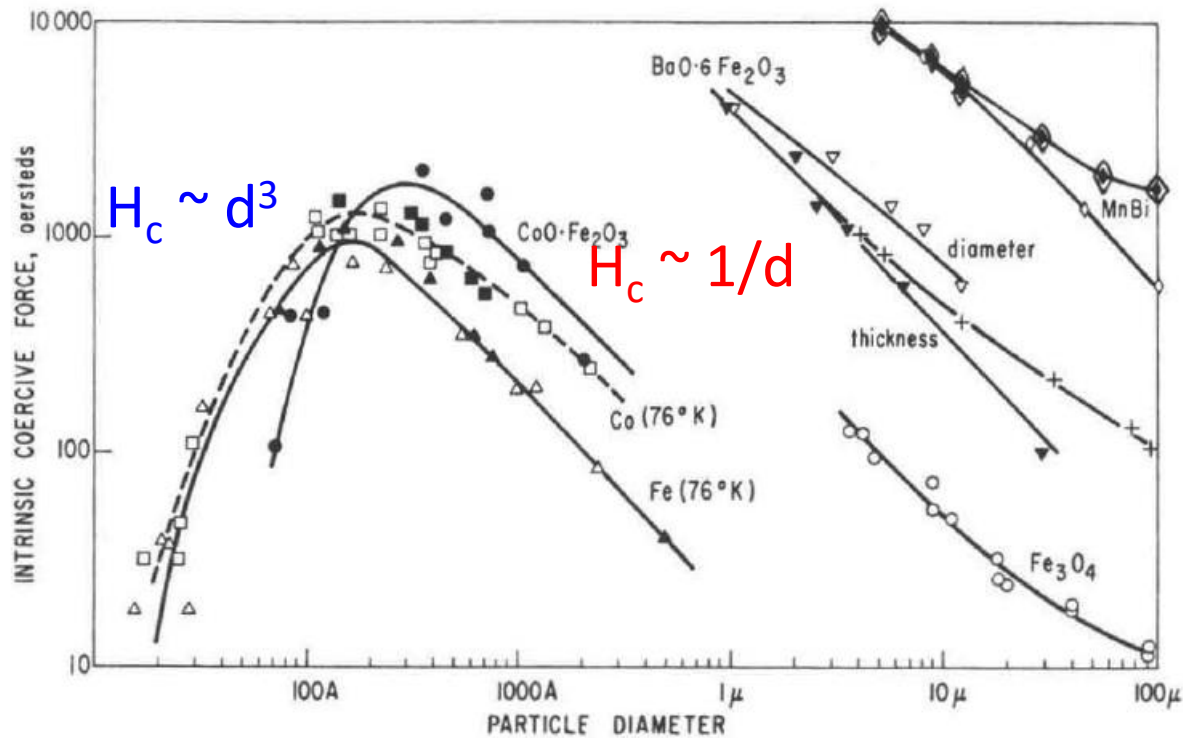


Skyrmions

# Nanomagnetism Outline

1. Ideal Monodomain magnetic nanoparticles
2. Real magnetic nanoparticles
  - a. surface effects
  - b. exchange bias
3. Interactions of magnetic nanoparticles
  - a. magnetostatic
  - b. exchange coupling (nanocrystalline alloys)
4. Magnetic Nanowires
5. Magnetic Nanopatterns
  - a. single domain
  - b. magnetic vortices
  - c. skyrmions

# Size Dependence of Coercivity in Fine Particles



Monodomain

Multidomain



Superparamagnetic,  $H_c = 0$

# 3 Key Parameters

**$M_s$  --- Saturation magnetization**

**A --- Exchange stiffness**

- $E_{\text{ex}} = |-2J S \cdot S| = 2 aA$ ,  $a$  = lattice spacing
- $k_B T_C = E_{\text{ex}}$

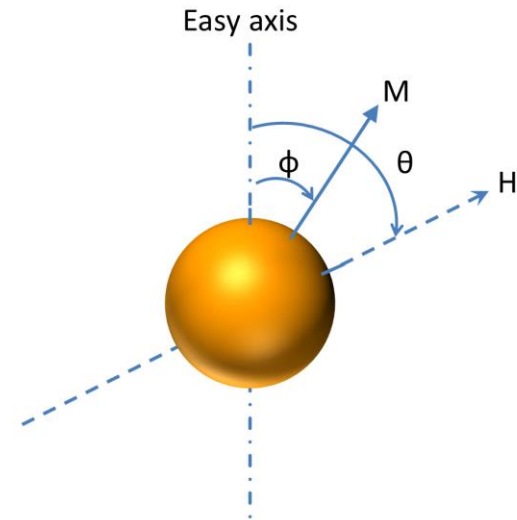
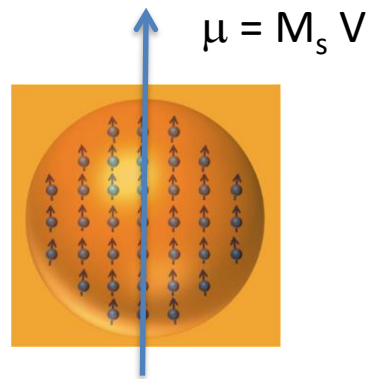
**K --- Anisotropy**

- Magnetocrystalline
- Shape (*0 for a spherical particle*)
- Magnetostriction (*ignore here*)
- Surface (*ignore for now*)

# Monodomain Particles

Monodomain at a size below  $d_{cr}$ :

$$d_{cr} = \frac{72(AK)^{1/2}}{m_0 M_s^2}$$



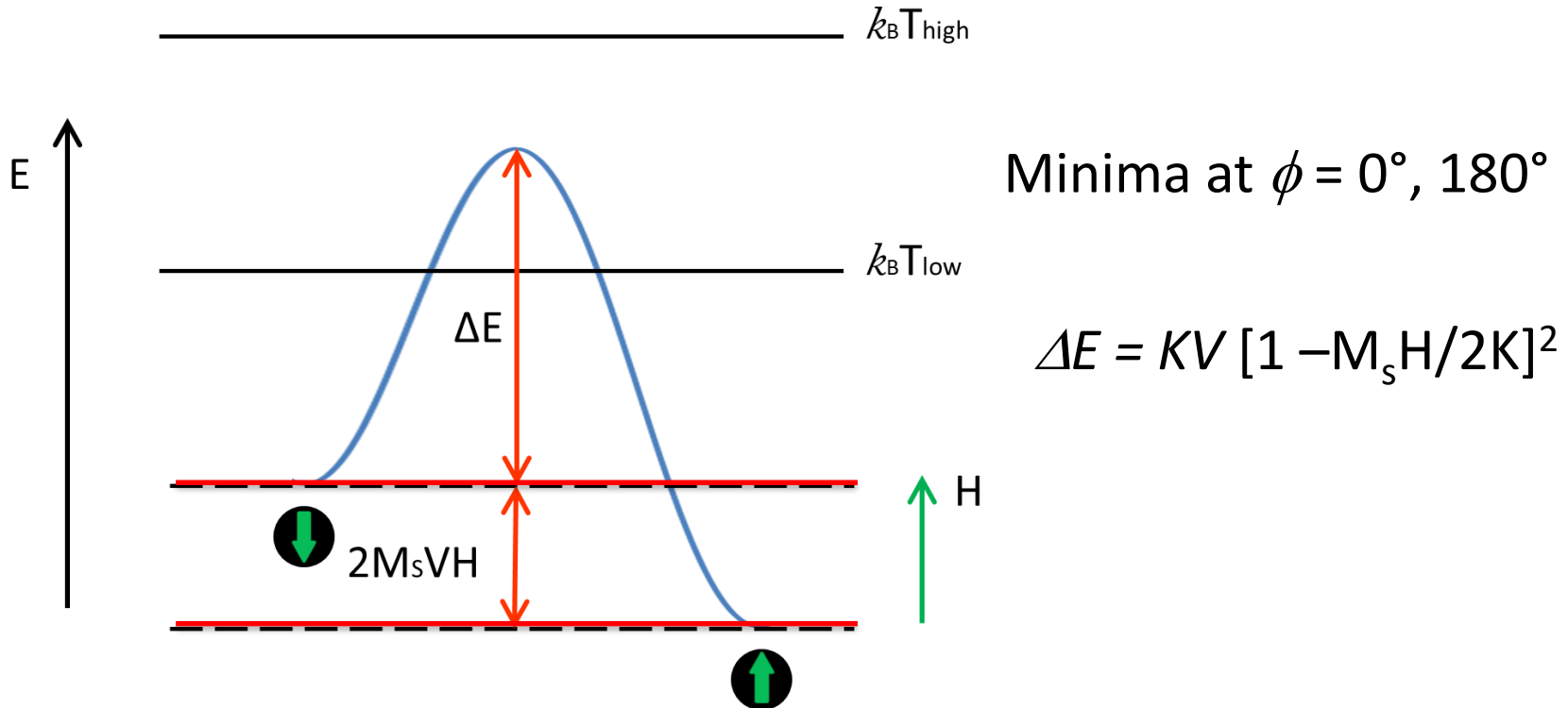
Assume uniform magnetization, treat as uniaxial

$$E = KV \sin^2 \phi - M_s V H \cos(\theta - \phi)$$

If  $H$  parallel to the Easy Axis, 2 energy minima:  $\phi = 0^\circ, 180^\circ$

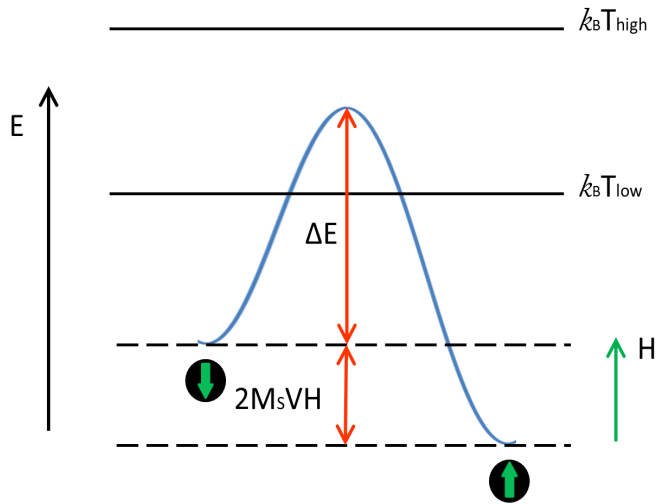


# Energy Barrier to Reversal



Since no domain walls, reversal occurs by **Coherent Rotation** of particle spins

# Energy Barrier Picture



Estimate rate of relaxation through Arrhenius law

$$\tau^{-1} = \tau_0^{-1} \exp(-\Delta E/k_B T)$$

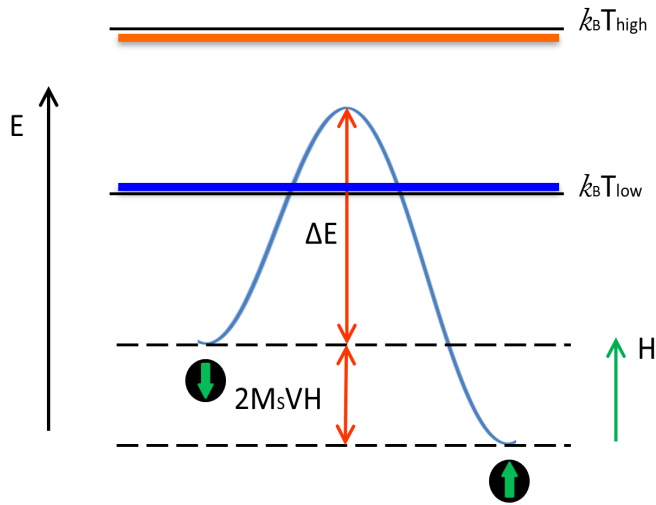
$$\tau^{-1} = \tau_0^{-1} \exp(-KV/k_B T)$$

where  $\tau_0^{-1}$  is the Larmor precession frequency,  $\sim 1$  GHz

Over time, the system will equilibrate

--- if this happens within the measurement time, the particles are said to be **Superparamagnetic**

# Superparamagnetism



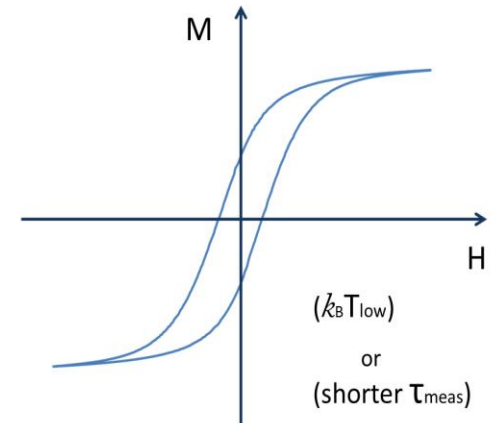
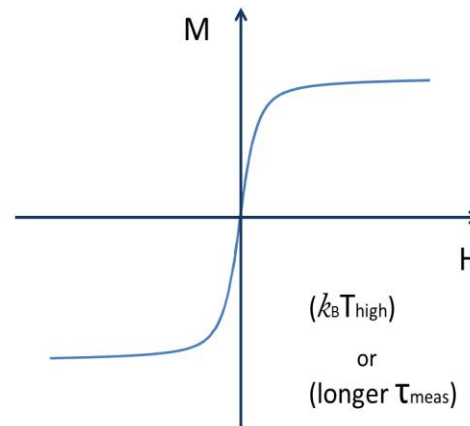
$$\tau^{-1} = \tau_0^{-1} \exp(-\Delta E/k_B T) = \tau_0^{-1} \exp(-KV/k_B T)$$

For **High temperature** relaxation is faster than at **Low temperature**

See no hysteresis if relaxes within measurement time

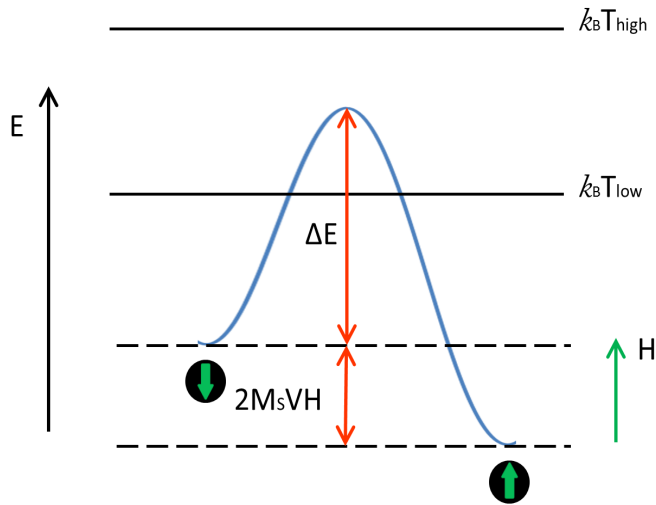
$$\tau < \tau_{\text{meas}}$$

$$d_{SP} = \frac{6kT_B}{\rho K \ln(t_{\text{meas}}/t_0)} \dot{u}^{1/3}$$



**Superparamagnetic**

# Checking for Superparamagnetism



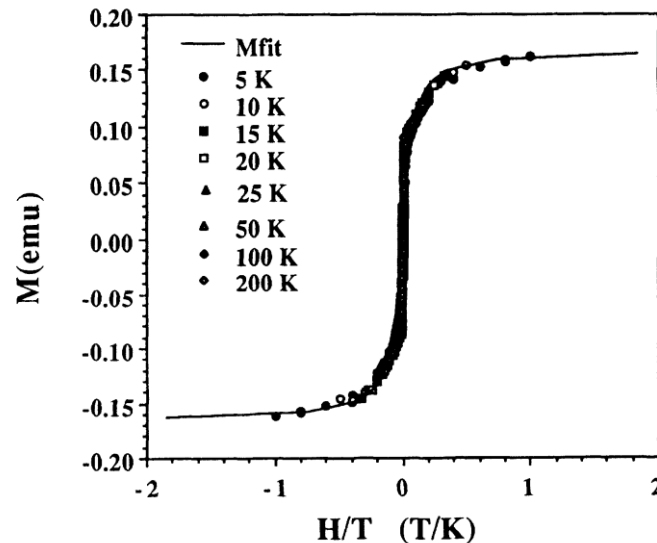
Estimate quasi-equilibrium populations based on Boltzmann factors

Langevin function,

$$L(x) = M/M_0 = \coth(x) - 1/x$$

$$\text{where } x = M_s V H / k_B T$$

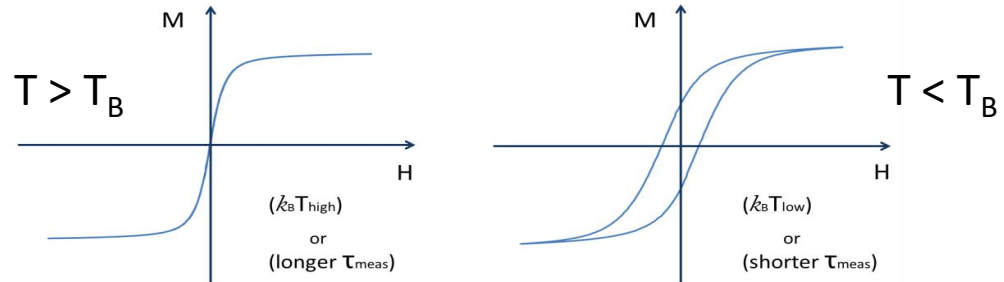
$M = M(H/T)$ , fit for dilute sample to determine particle moment  $\mu = M_s V$



# Blocking Temperature

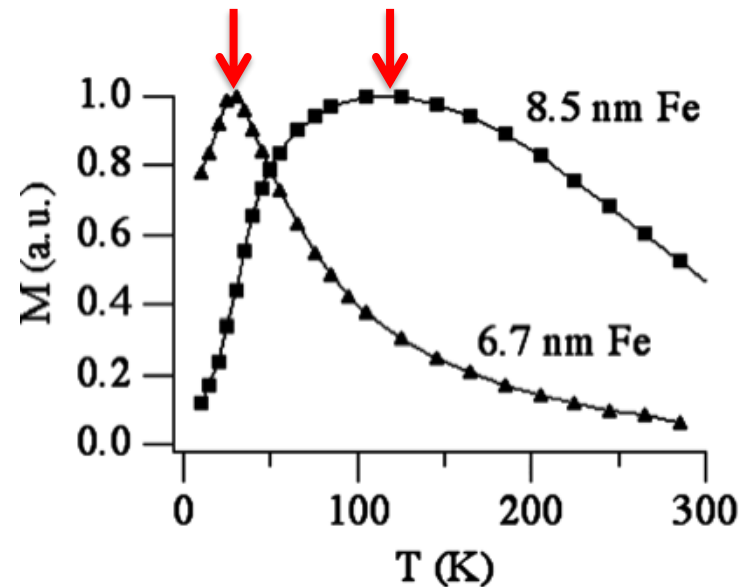
The coercivity  $H_c$  goes to zero at the Blocking temperature,  $T_B$

$$T_B = KV \ln(t_{meas} / t_0) / k$$

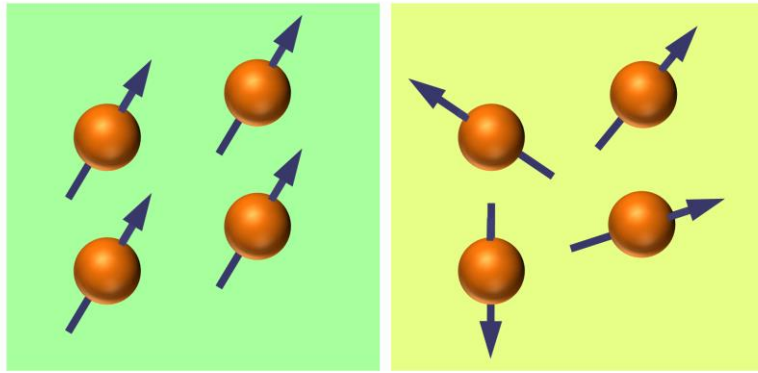


Find  $T_B$  from the peak in the Zero Field Cooled Magnetization,  $M_{zfc}(T)$

- cool with  $H = 0$
- turn on small  $H$ , measure  $M$  as increase  $T$



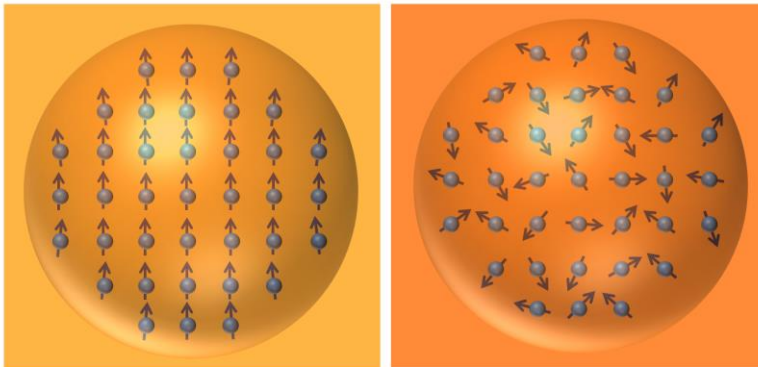
# Blocking vs. Curie Temperature



$T < T_B$

$T > T_B$

Hysteresis is seen below the **Blocking temperature** because the *particle spins* haven't yet randomized

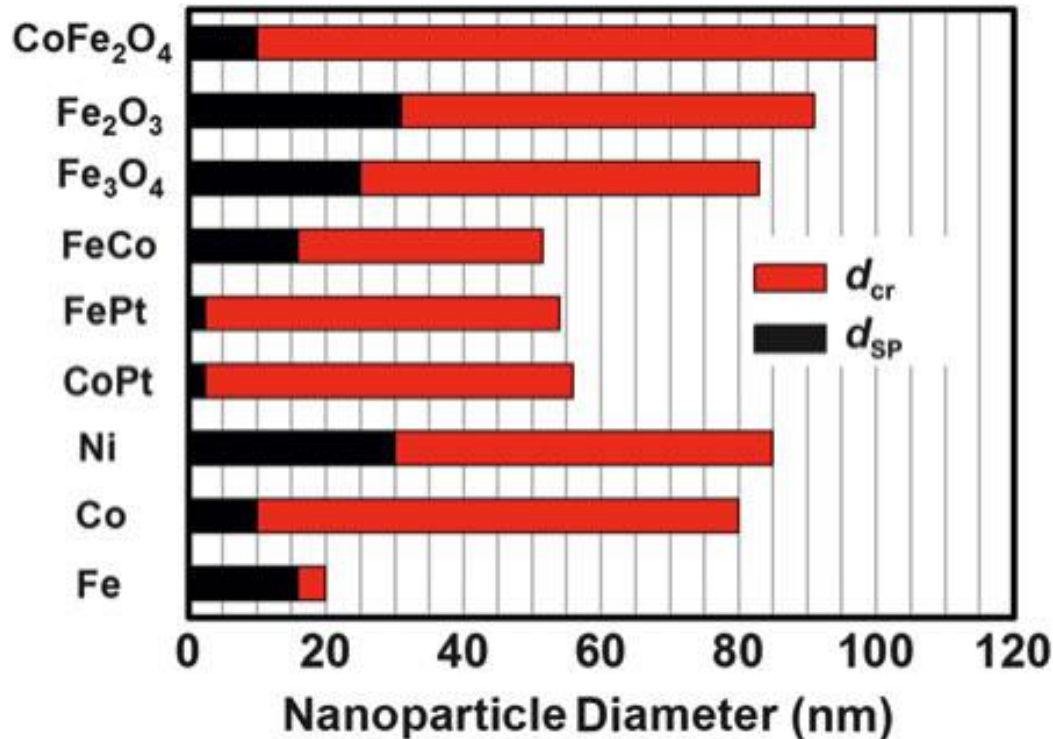


$T < T_C$

$T > T_C$

Hysteresis is seen below the **Curie temperature** because the *atomic spins* are not randomly oriented

# Monodomain and Superparamagnetic Thresholds for Different Materials



For soft magnetic materials (low  $K$ ),

$$d_{sd} \sim l_{ex} \ll \text{domain wall width } \delta$$

For hard magnetic materials (big  $K$ ),

$$d_{sd} \gg l_{ex} \geq \text{domain wall width } \delta$$

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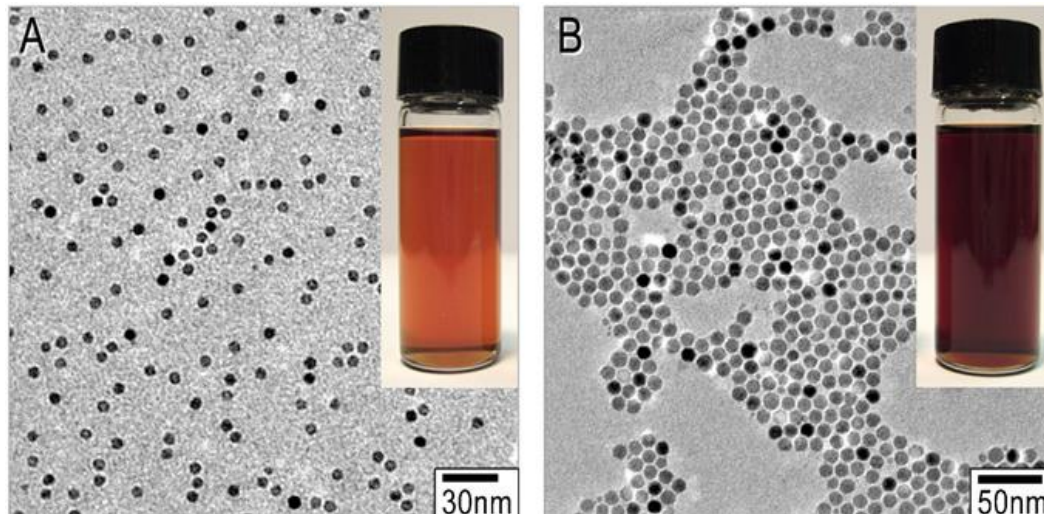
# What Could Be Different for Real Nanoparticles?

**$M_s$  --- Saturation magnetization** --- Could be reduced if there is surface oxide (e.g., Co/CoO), if there are antiphase boundaries (e.g., in  $\text{Fe}_3\text{O}_4$ ), or if there is surface disorder

**A --- Exchange stiffness** --- Fewer near neighbors reduces  $E_{\text{ex}}$ , symmetry breaking at a surface or interface could enable noncollinear exchange (Dzyaloshinskii-Moriya interaction, DMI)

**K --- Anisotropy** --- Symmetry breaking and/or spin disorder at the surface modify the anisotropy

# Thermal Decomposition in High Boiling Point Organic Solvents



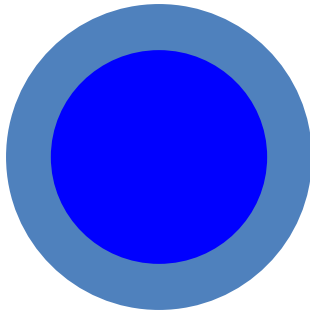
More monodisperse, high crystallinity

S. H. Sun, et al., J. Am. Chem. Soc. **126** 273 (2004). --- ~ **84 emu/g**

J. Park, et al. Nat. Mater. **3**, 891-895 (2004). --- ~ **42 emu/g but make gram quantities**

# Reduced $M_s$ in Nanoparticles

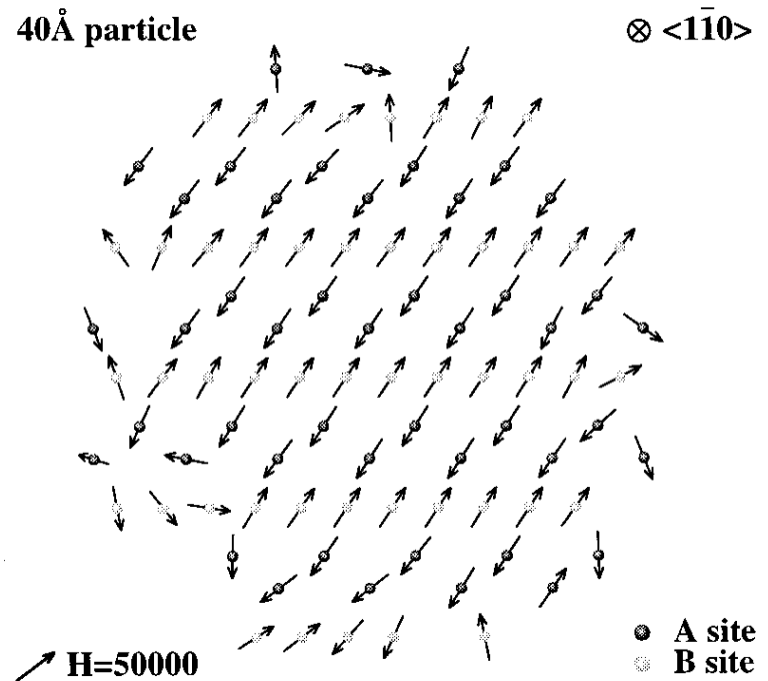
Magnetization less than expected based on the #spins



“Magnetic Dead Layer” = ?

R. H. Kodama and A. E. Berkowitz, *Phys. Rev. B* **59**, 6321 (1999).

## Surface Spin Glass

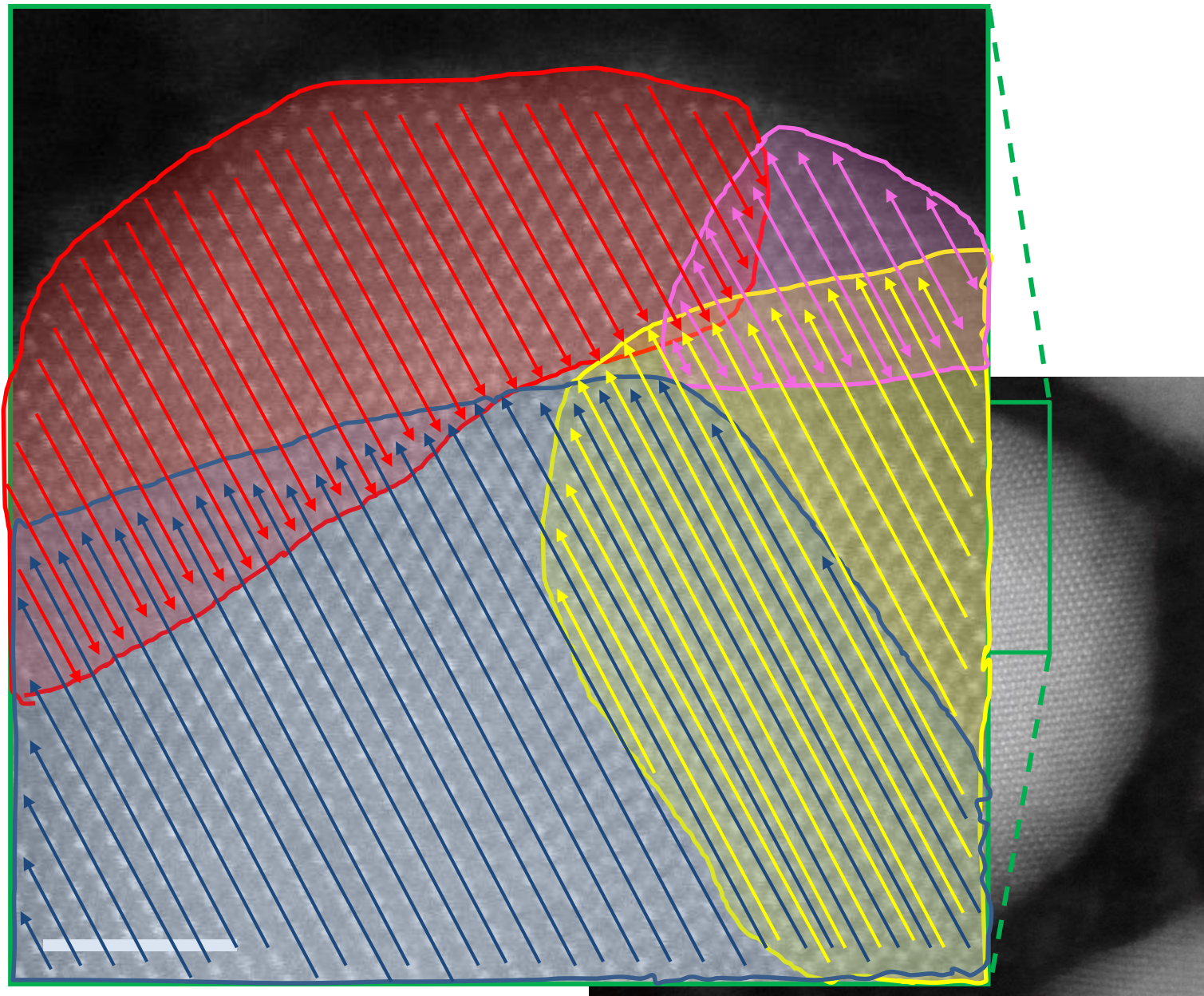


Model of  $\text{NiFe}_2\text{O}_4$  made by ball milling

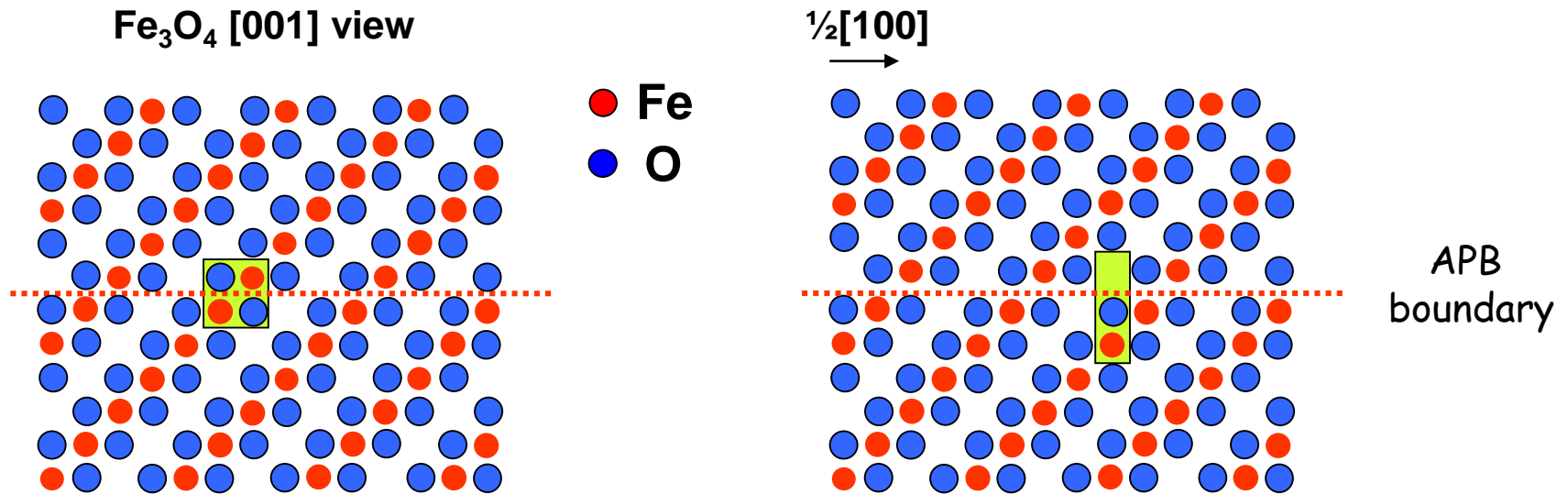
# STEM-HAADF of Iron Oxide with APBs

Chemically ordered with extensive structural defects

Structural domains separated by anti-phase boundaries



# Exchange Interactions Across Antiphase Boundaries in $\text{Fe}_3\text{O}_4$



Super exchange interactions in bulk

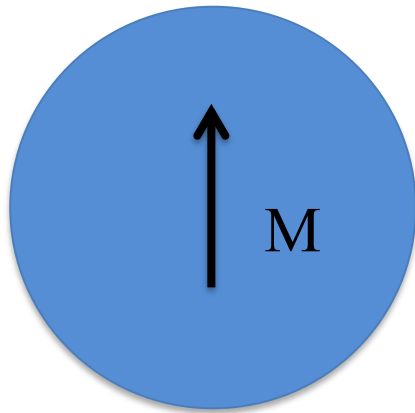


Super exchange interactions across APB



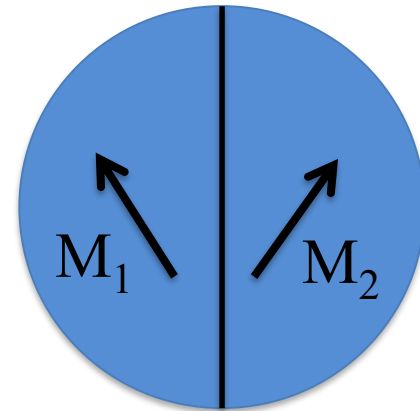
# Effect of Antiphase Boundaries on Spin Configuration

No Antiphase Boundary



Monodomain

With Antiphase Boundary



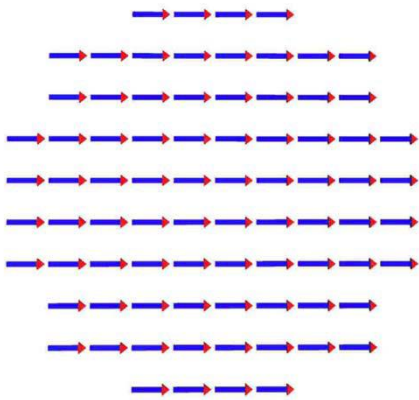
Not monodomain even though below  $d_{cr}$

Lower  $M_s$  and acts like higher  $K$

[Z. Nedelkoski, et al., *Sci. Rep.* **7**, 45997 (2017)]

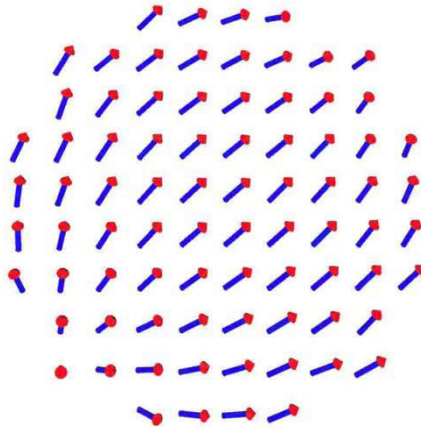
# Effect of Surface Anisotropy $K_s$

Monodomain



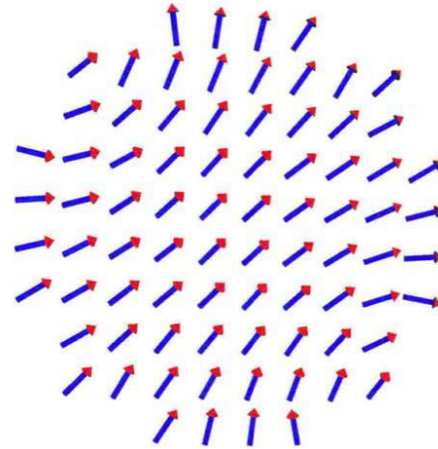
$$K_s = 0$$

Artichoke



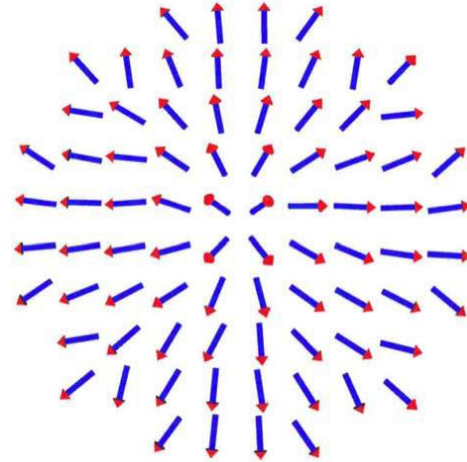
$$K_s < 0$$

Throttled



$$K_s > 0, \\ \text{small}$$

Hedgehog



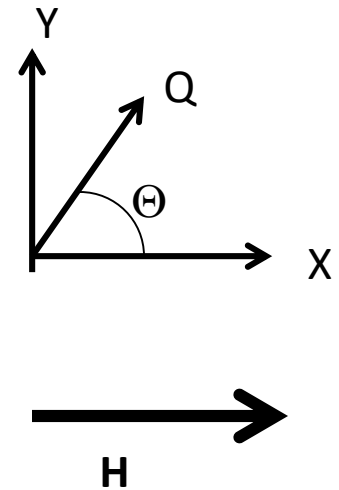
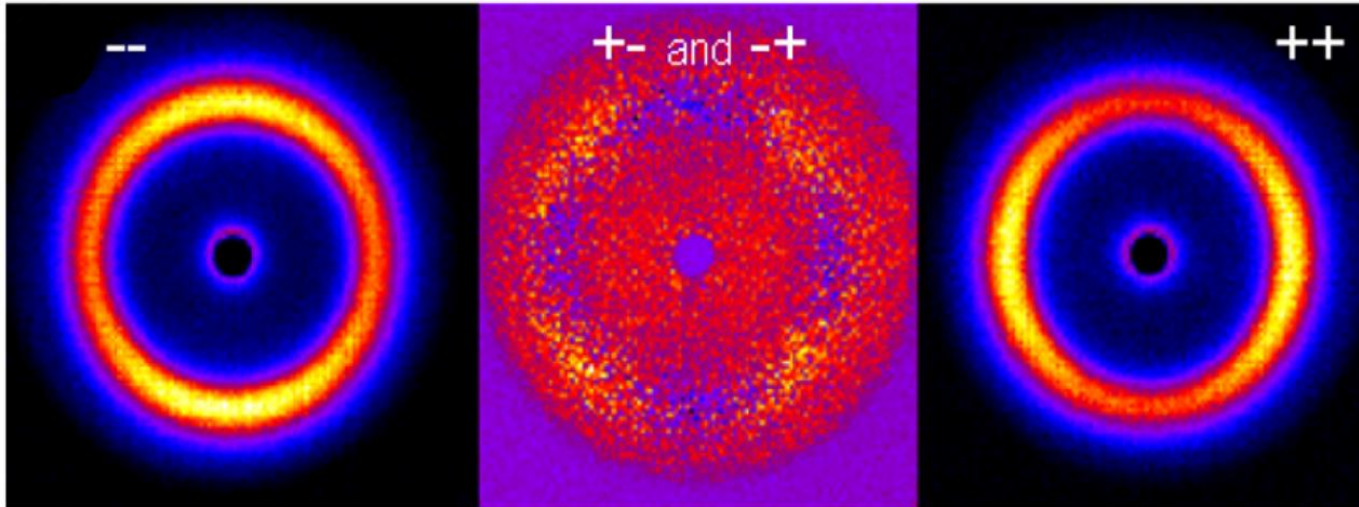
$$K_s > 0, \\ \text{large}$$

L. Berger, et al., *Phys. Rev. B* **77**, 104431 (2008).

Very difficult to determine  $K_s$  experimentally

# Polarized Small Angle Neutron Scattering (SANS)

9.0 nm  $\text{Fe}_3\text{O}_4$  nanoparticle crystals, 1.2 T

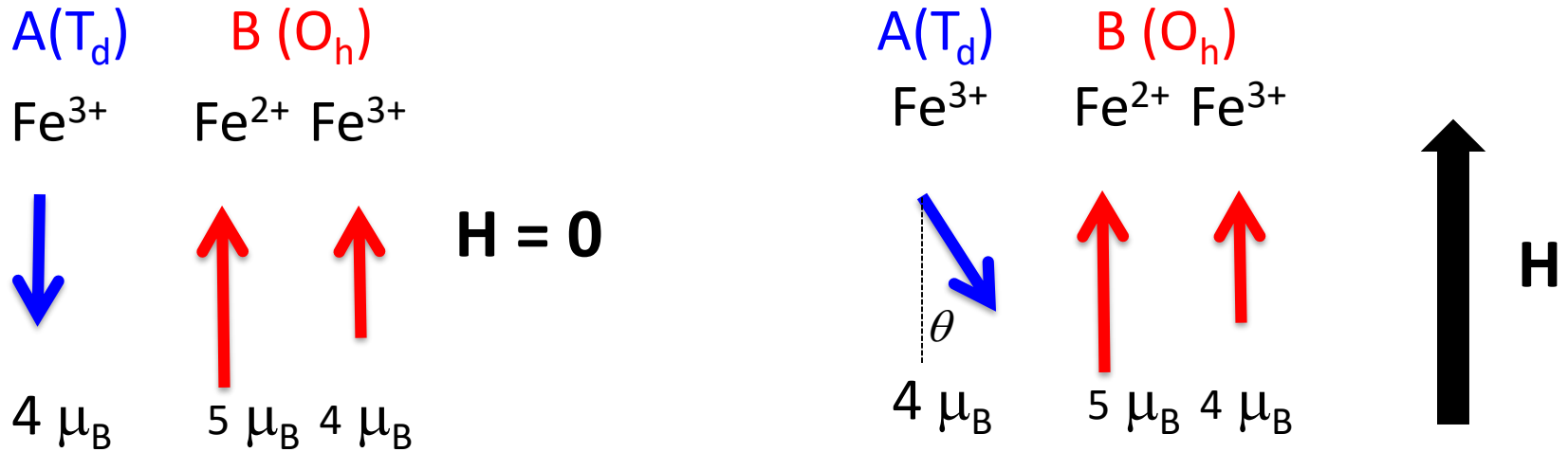


Unequivocally differentiate magnetic and nuclear scattering  
- unlike with previously where relied on a “saturating” field

**$M_x \neq M_y \neq M_z$ , and see unexpected evidence of perpendicular M**



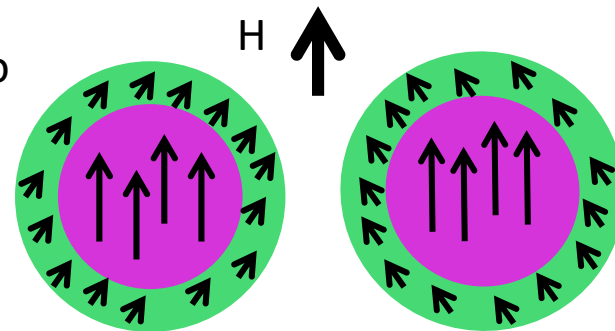
# Spin Canting in $\text{Fe}_3\text{O}_4$ NPs



Ferrimagnetic

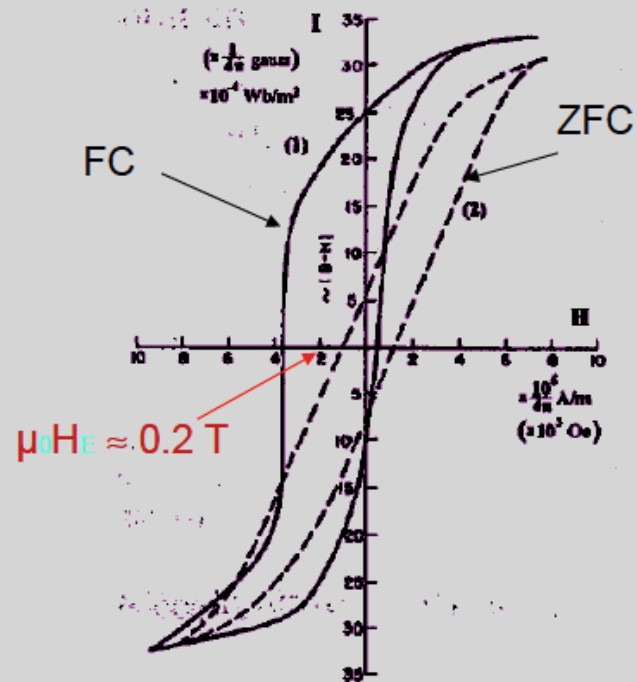
Zeeman interactions compete with exchange

Enhanced for spins near surface due to weaker exchange



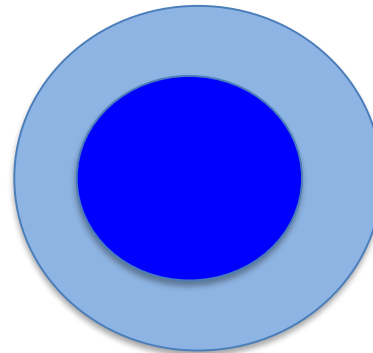
# Exchange Bias in Core-Shell Particles

Oxidized Co nanoparticles



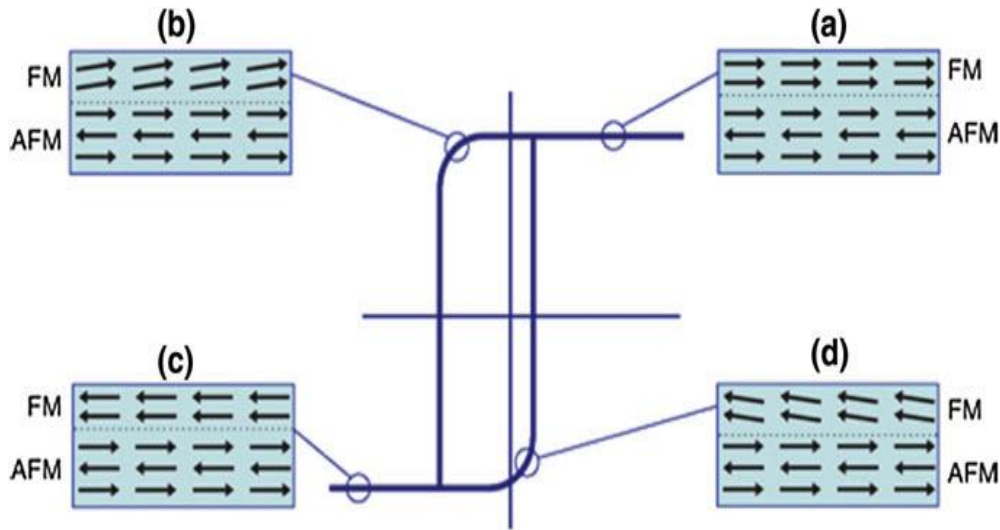
Meiklejohn and Bean,  
Phys. Rev. 102, 1413 (1956),  
Phys. Rev. 105, 904, (1957)

See difference in  $H_c$  and a loop shift if NPs are cooled in a magnetic field below the Neel temperature of CoO



Co (FM) core and CoO (AF) shell

# Exchange Bias in Core-Shell Particles

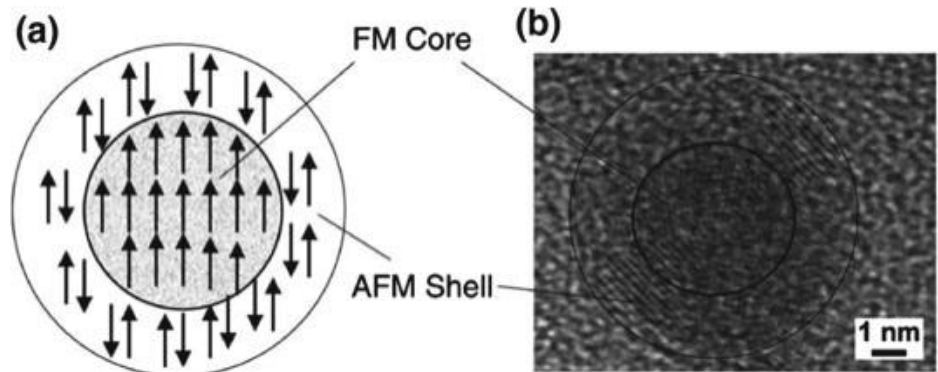


Field cooled loop

**Training effects:** the FM/AF interface spin configuration depends on H, time, and temperature

(A. Guimaraes, Nanomagnetism)

Exchange bias more complex in particles than thin films because shell is polycrystalline



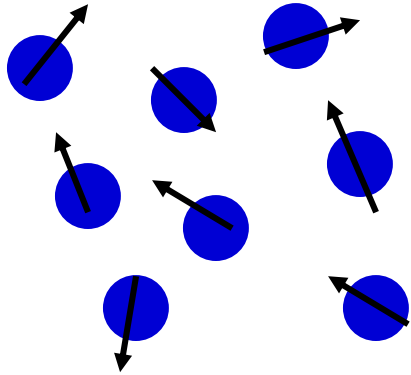
TEM image

# Nanomagnetism Outline

1. Ideal monodomain magnetic nanoparticles
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  - a. surface effects
  - b. exchange bias
3. Interactions of magnetic nanoparticles
  - a. magnetostatic
  - b. exchange coupling (nanocrystalline alloys)
  - c. exchange coupling (exchange spring composites)
4. Magnetic Nanowires
5. Magnetic Nanopatterns
  - a. single domain
  - b. magnetic vortices
  - c. skyrmions

# Magnetostatic Interactions

Superparamagnet



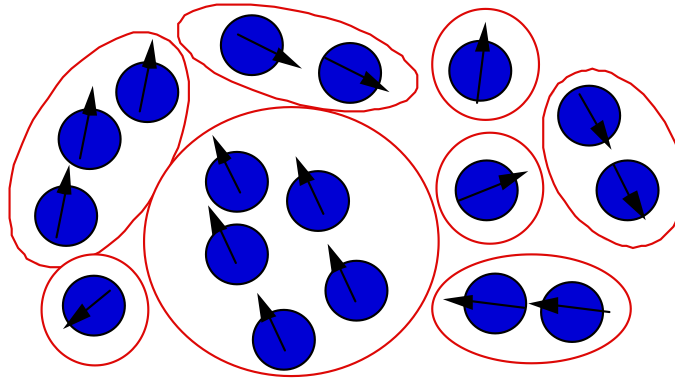
No interactions between particles

$M \sim$  Langevin function

$M(t)$  decays exponentially

**Dilute**

Superferromagnet



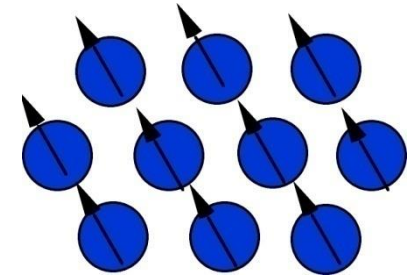
Some interactions

Short-range magnetic order

Glassy  $M(t)$  behavior

**> 1 vol.%, but  
structurally disordered**

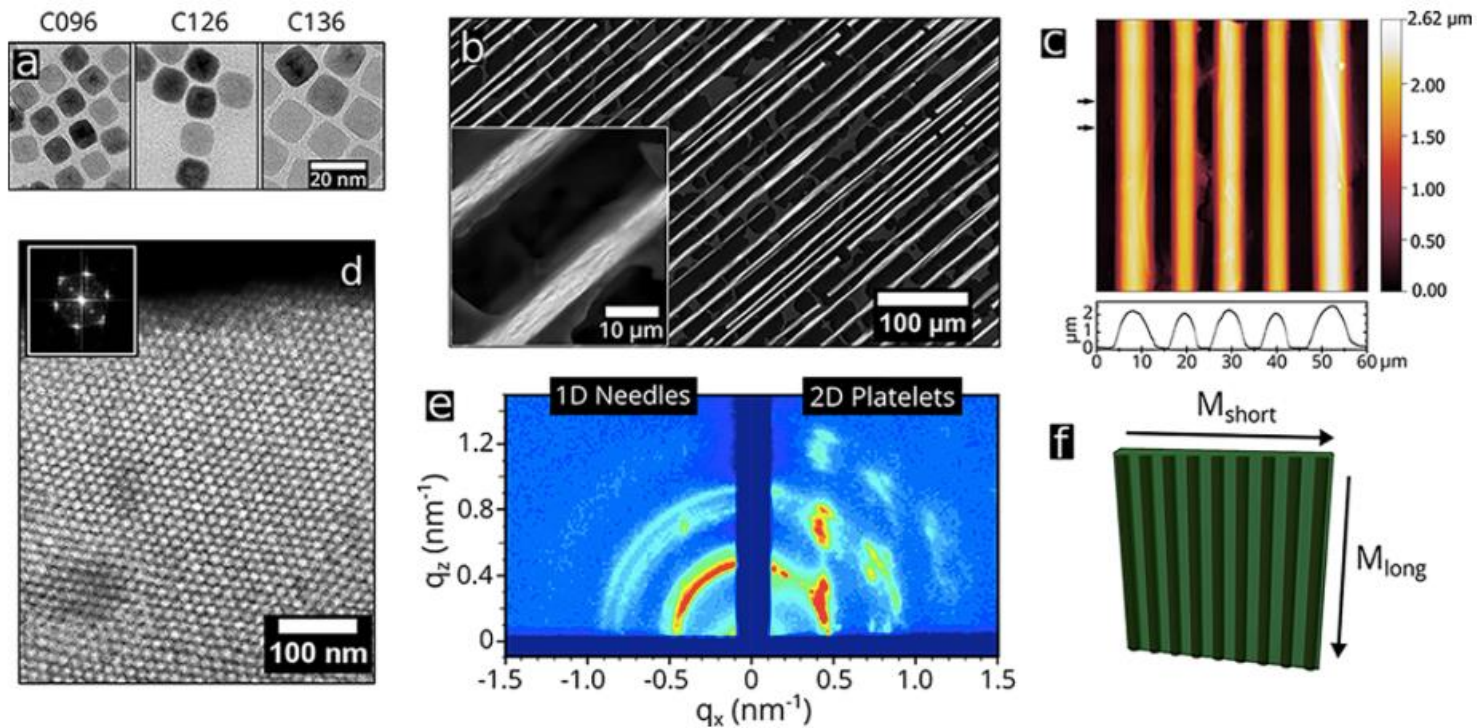
Dipolar Ferromagnet



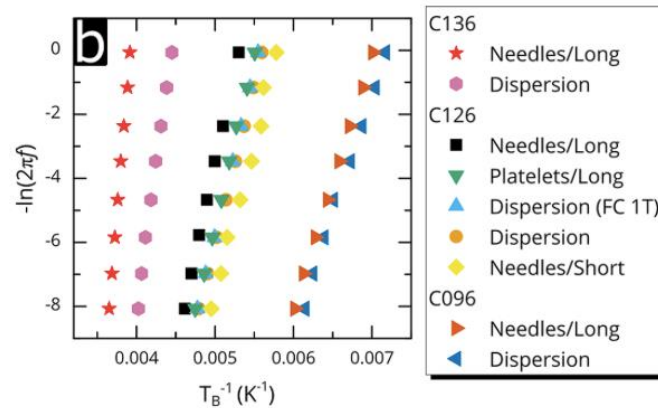
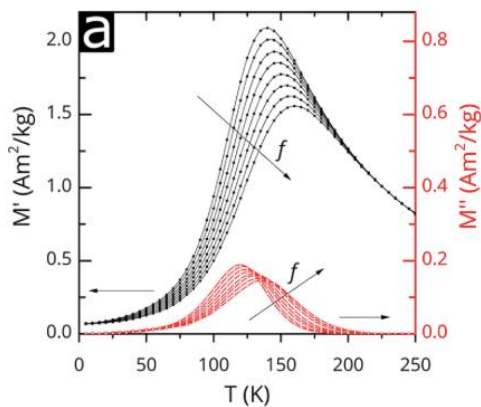
Long-range magnetic order that is stable over time

**Highly ordered**

# Collective Behavior in Nanoparticle Assemblies with Dipolar Interactions



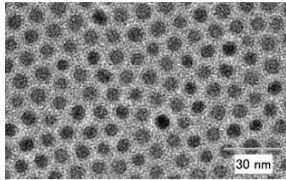
# AC Magnetic Response in Nanoparticle Assemblies and Superferromagnetism



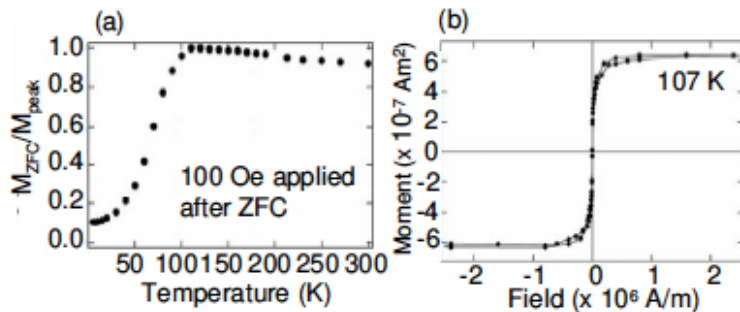
Magnetostatic interactions can increase the Blocking temperature, stabilizing multi-particle spin correlations

**Superferromagnetism:** Ordering of spins over neighboring particles, and collective switching due to thermal fluctuations

# Magnetic Domains Observed by Electron Holography

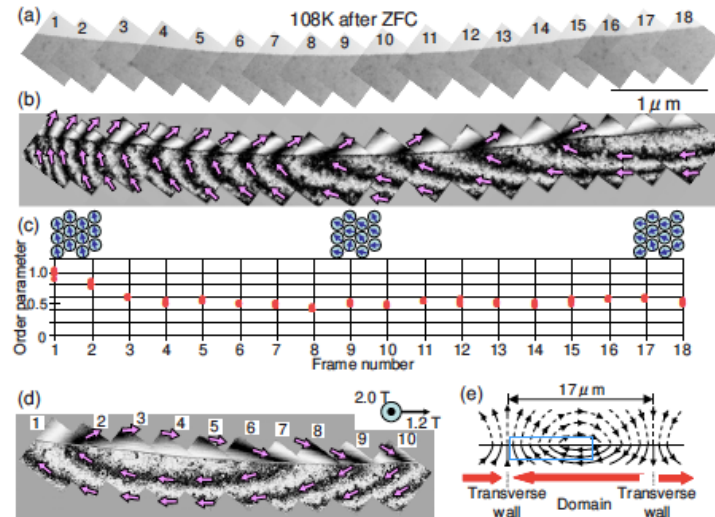


8 nm Co nanoparticle monolayer



Seems like it should be superparamagnetic, but it's not

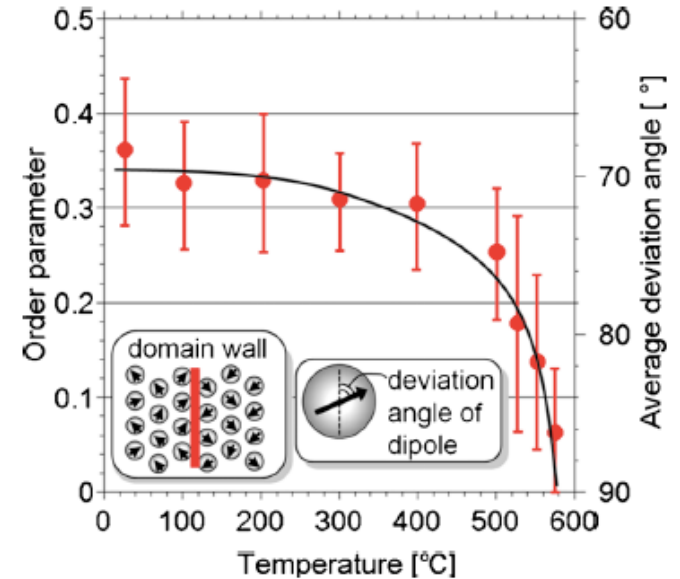
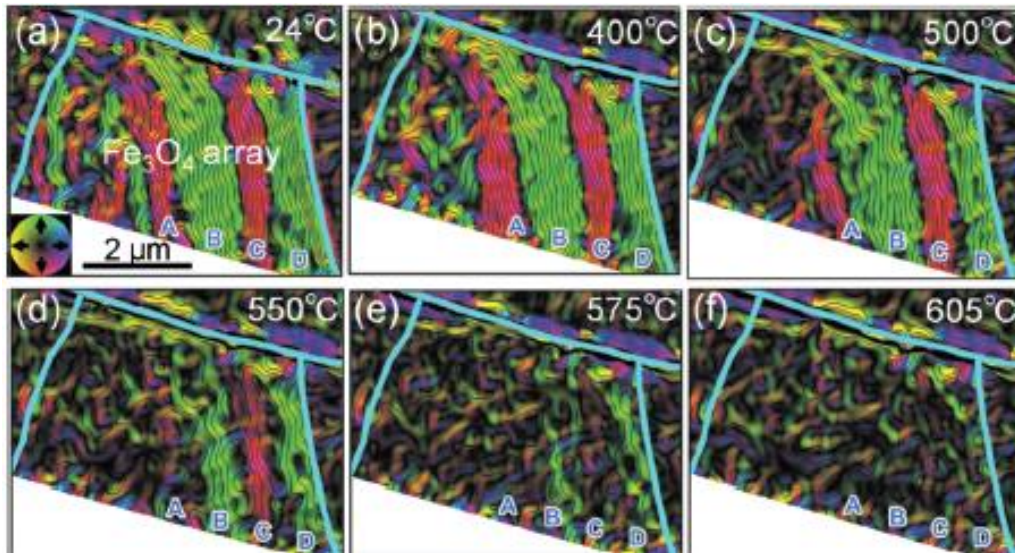
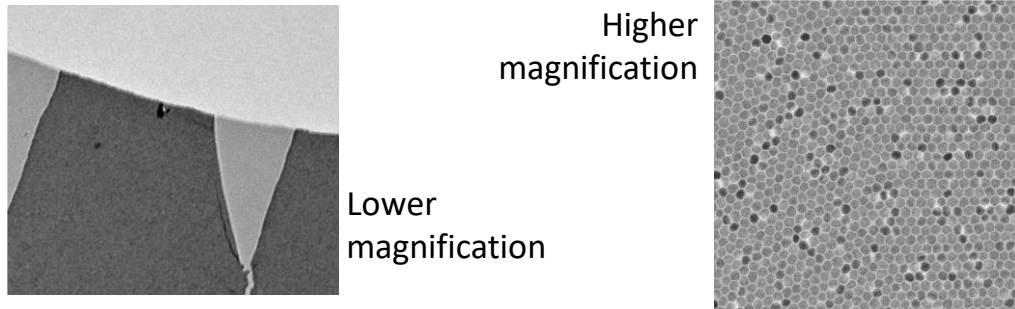
**Assembly acts like a soft ferromagnet in DC measurements, with lower  $K_{\text{eff}}$  due to anisotropy averaging**



Spontaneous magnetization is stable over time, so ferromagnetic, but only dipolar interactions between particles



# Dipolar Ferromagnetism in 13 nm $\text{Fe}_3\text{O}_4$ Nanoparticle Assemblies



Looks like  $M(T)$  for a ferromagnet



Electron Holography phase shift reveals in-plane B

K. Yamamoto, et al., Appl. Phys. Lett. **98**, 072509 (2011).

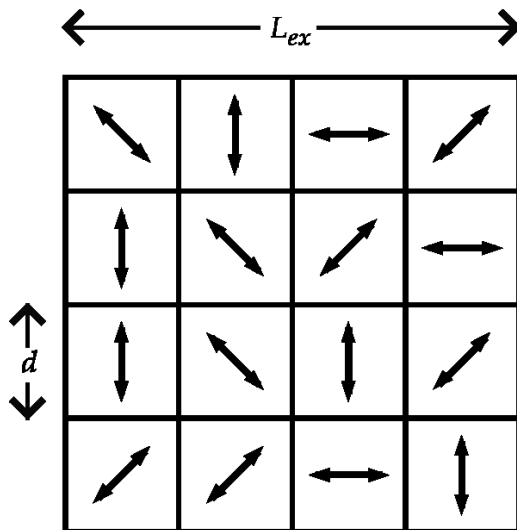
# What About Exchange Coupling?

| Length                              | Symbol                | Typical Magnitude (nm) |
|-------------------------------------|-----------------------|------------------------|
| Exchange length                     | $l_{\text{ex}}$       | 1 - 100                |
| Critical single domain diameter     | $d_{\text{cr}}$       | 10 – 1000              |
| Critical superparamagnetic diameter | $d_{\text{sp}}$       | 1 – 100                |
| Spin diffusion length               | $\lambda_{\text{sd}}$ | 1 – 100                |
| Domain wall width                   | $\delta$              | 1 – 100                |

# Exchange Interactions in Soft Nanocrystalline Materials

No surfactant between grains, make by melt spinning

$N$  touching grains of size  $d$   
with different orientations



$$N = \left( \frac{L_{ex}}{d} \right)^3$$

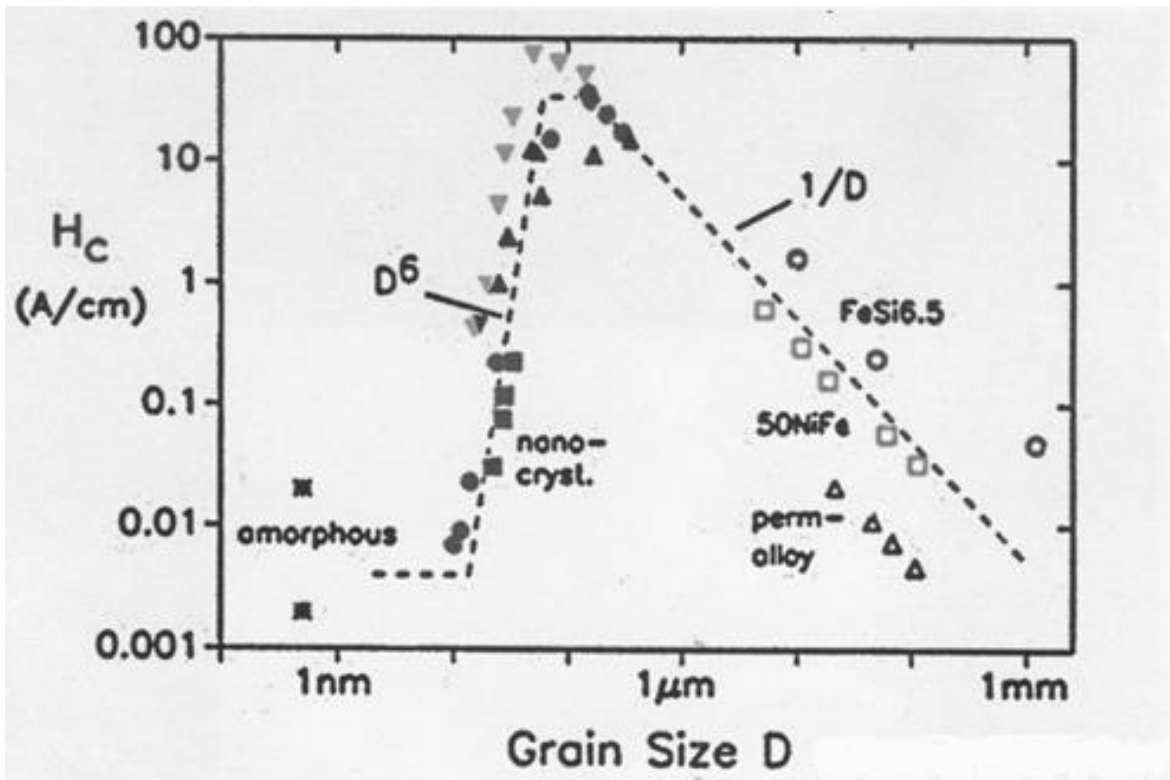
$$L_{ex} = \sqrt{\frac{2A}{m_0 M_s^2}}$$

$$\langle K \rangle \approx \frac{K}{\sqrt{N}}$$

Reduced  
anisotropy

When exchange length  $>$  grain size,  
and grains have different easy axes,  
get anisotropy averaging

# Coercivity in Soft Nanocrystalline Materials



Stronger size dependence than monodomain nanoparticles

$$H_c \sim D^6 \text{ vs } D^3$$

Reduced  $H_c$  since reduced  $K$

G. Herzer, IEEE Trans. Magn. **26** 1397 (1990)

Finemet:  $\text{Fe}_{73}\text{Si}_{13.5}\text{B}_9\text{Nb}_3\text{Cu}_1$

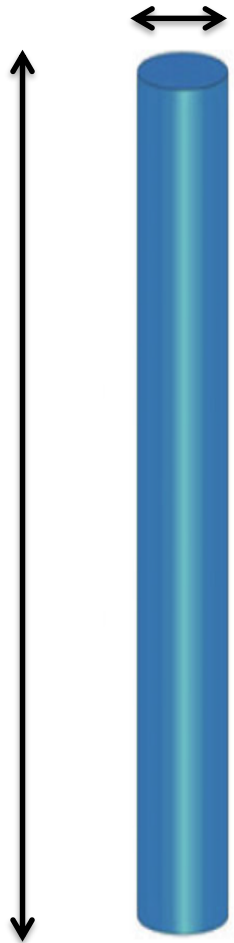
Nanoperm:  $\text{Fe}_{89}\text{Zr}_7\text{B}_3\text{Cu}_1$

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# Magnetic Nanowires

10 – 200 nm



>10  $\mu\text{m}$

Because of the large shape anisotropy, they can ideally be single domain

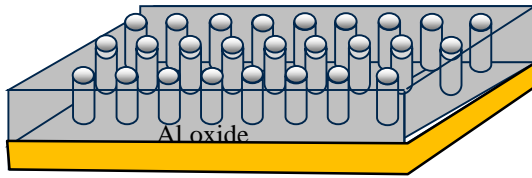
Real nanowires often have surface effects and magnetostatic interactions, just like nanoparticles

Because there is more magnetic volume, they are unlikely to be superparamagnetic

They do not switch their magnetization direction by Coherent Rotation; domain walls nucleate and travel down the length of the wire

# Making Nanowires

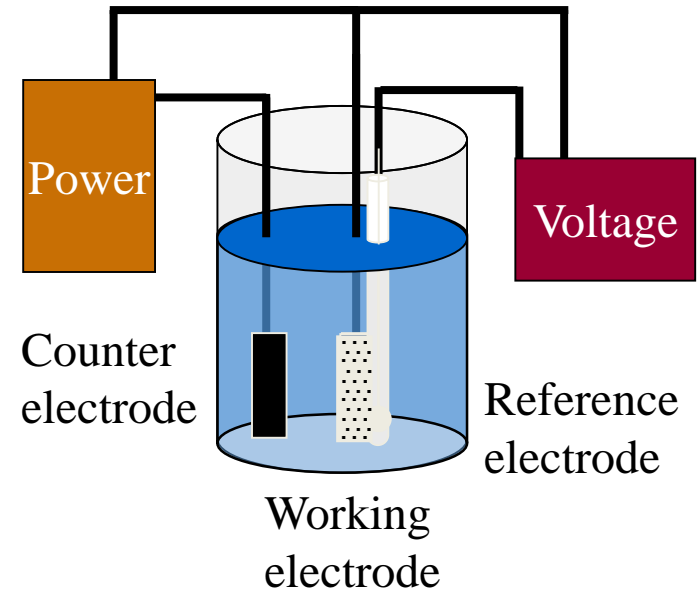
- **Make nanoporous template by anodizing aluminum:**



Schematic:

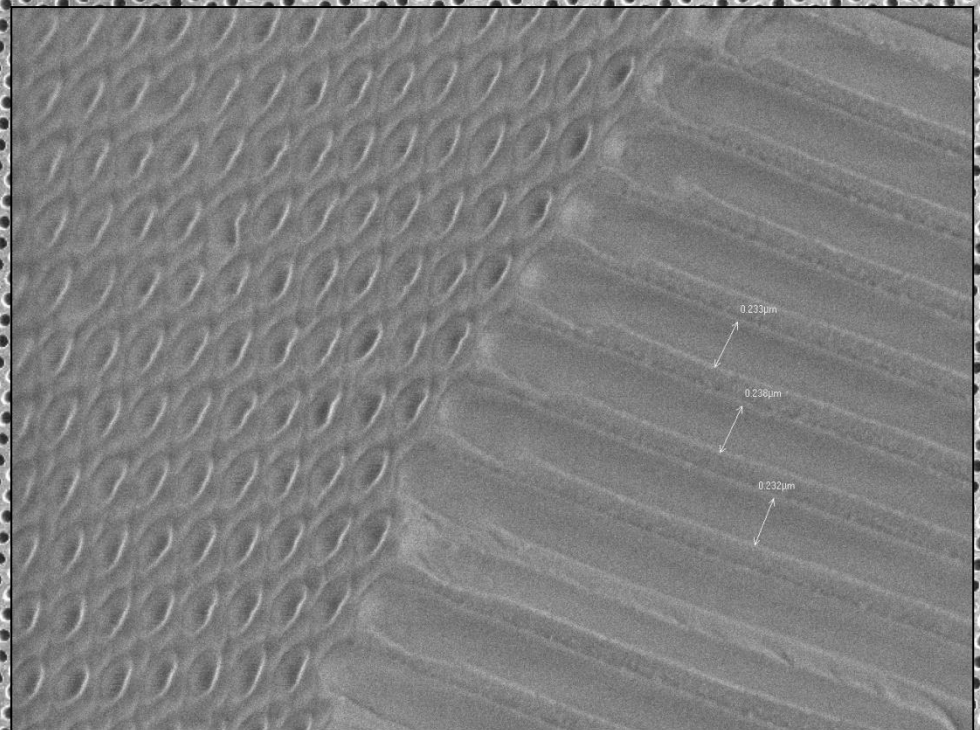
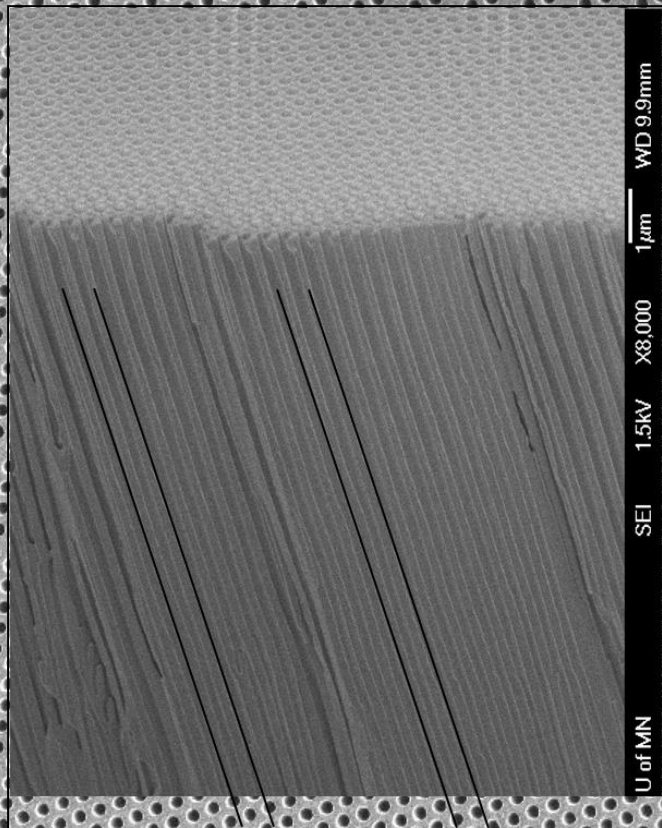
10-200nm pores can be achieved

- **Contact film on bottom template**
  - Can pattern contact for anisotropic arrays
- **Electroplate nanowires into pores**
  - Any conductive material can be made
- **Etch template to release wires if application requires**



(Bethanie Stadler, U. Minnesota)

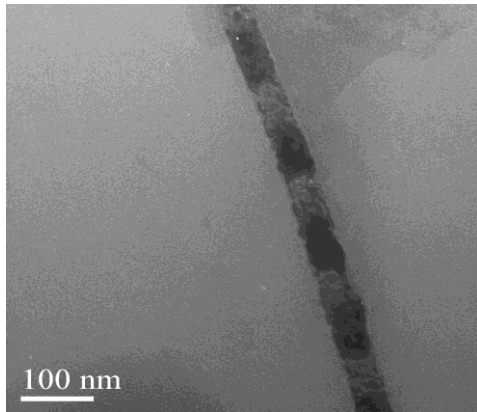
# SEM of Nanowire Templates



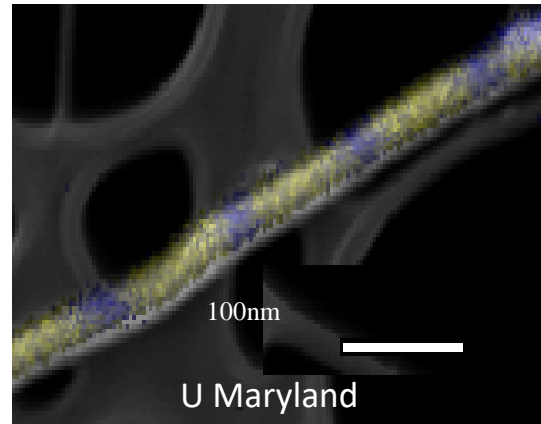
Maqablah et al. *IEEE Trans. Mag.* **48** (2012).  
Sung et al. *IEEE Trans. Mag.* **50** (2014).



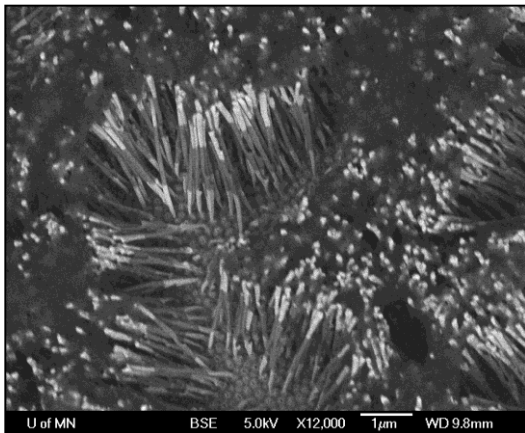
# Can Fill Pores with: Co, Cu, Ni, FeGa, Au... any metal?



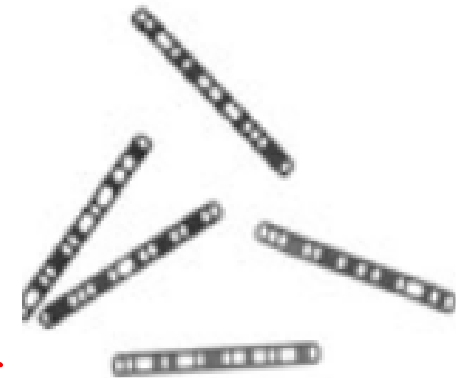
TEM of  $[\text{Co/Cu}]_n$  wire with  
**Giant Magnetoresistance (GMR)**



TEM/EELS of  
 $[\text{Cu/FeGa}$   
 $(\text{Galfenol})]_n$  wire --  
- **apply force**  
**through**  
**magnetostriction**



SEM of  $[\text{Au/Ni/Au}]_n$   
wires where **Au** is  
**used to enable**  
**selective attachment of**  
**biomolecules**



(Bethanie Stadler, U. Minnesota)

# Nanomagnetism Outline

1. Ideal monodomain magnetic nanoparticles
2. Real magnetic nanoparticles
  - a. surface effects
  - b. exchange bias
3. Interactions of magnetic nanoparticles
  - a. magnetostatic
  - b. exchange coupling (nanocrystalline alloys)
4. Magnetic Nanowires
5. Magnetic Nanopatterns
  - a. single domain
  - b. magnetic vortices
  - c. skyrmions

# 2D Nanopatterns

Most nanopatterns are formed lithographically from thin films



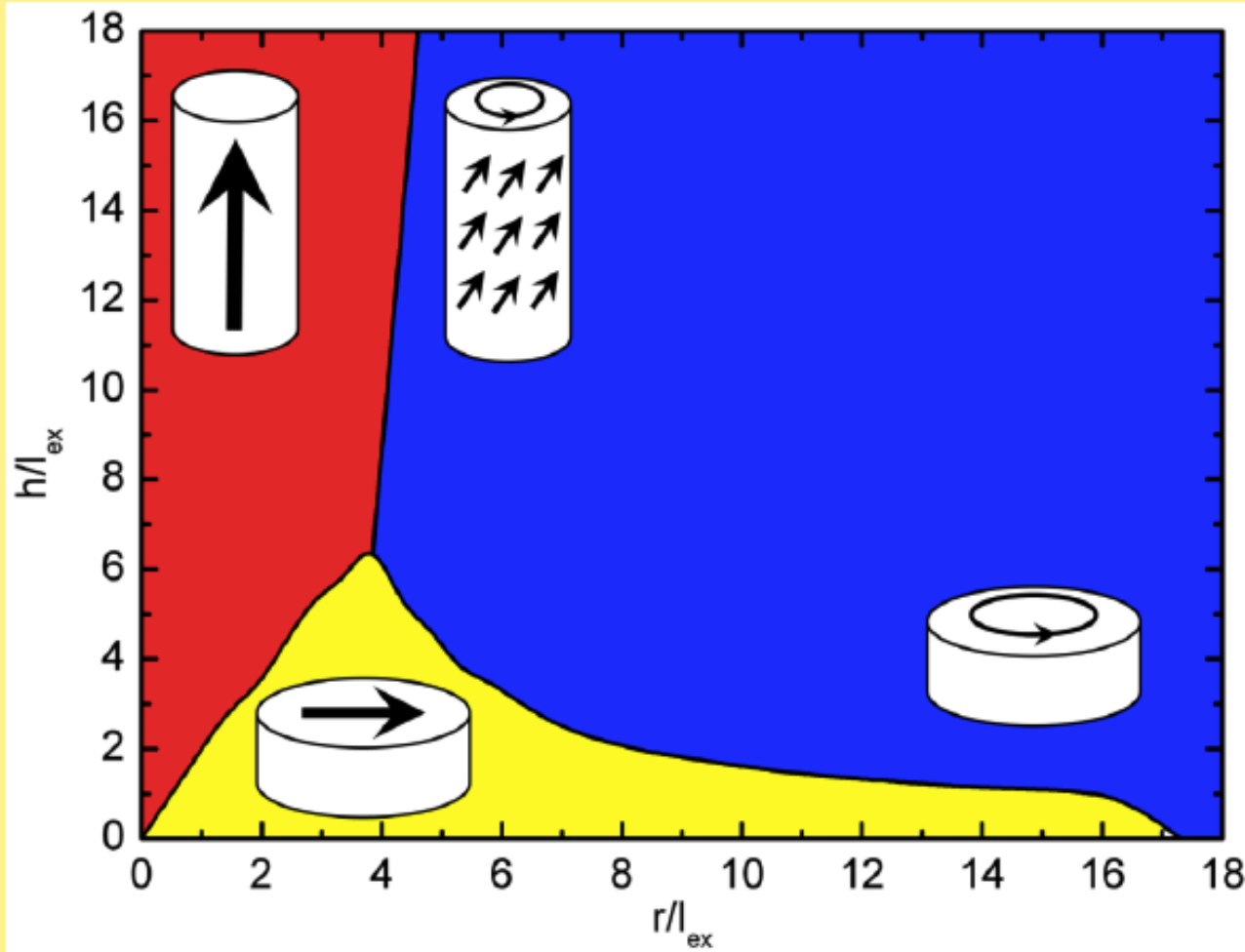
Can have crystallographic orientation, multilayer structures, and epitaxy with substrate (good for spintronics)



More recently, 2D spin configurations called skyrmions have been found that are stable without a confining pattern



# Disks can be monodomain or vortices, depending on the dimensions and $L_{ex}$



$$L_{ex} = \sqrt{\frac{2A}{m_0 M_s^2}}$$

Different spin configurations

Graph:  
Height vs. thickness

# Thin Circular Discs

There is shape anisotropy

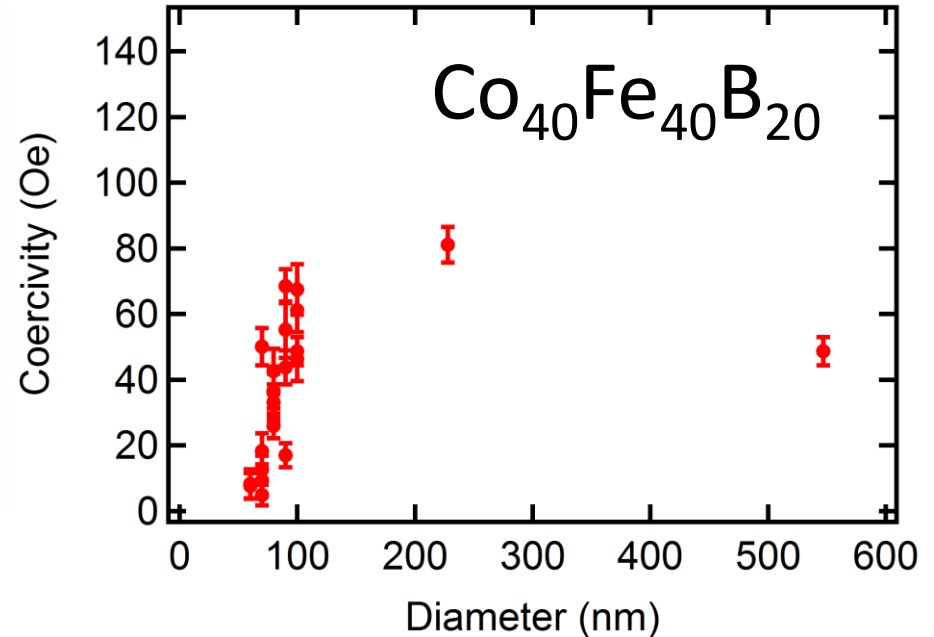
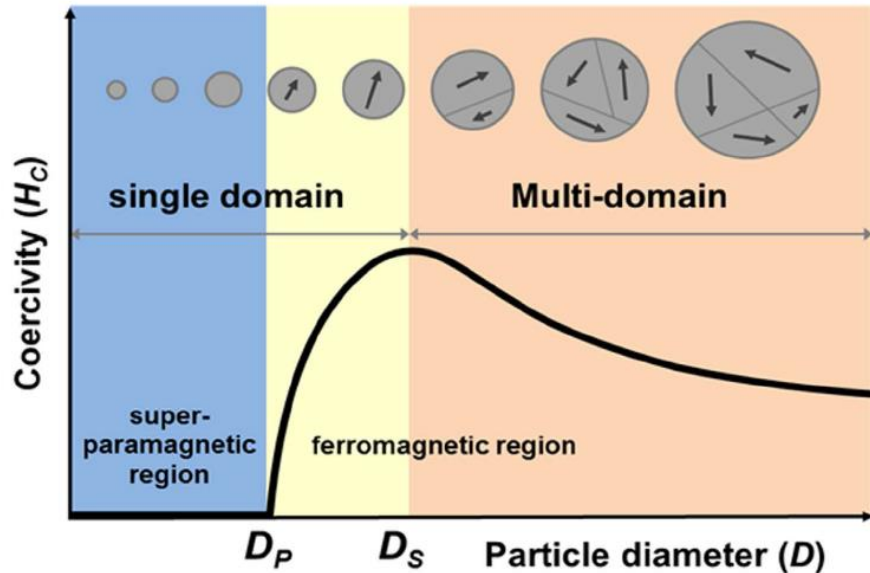
They can be stable in a monodomain state, but don't reverse their magnetization by Coherent Rotation except  $< 10$  nm and  $< 10$  K

They can be superparamagnetic for small diameters and thicknesses

For larger diameters a vortex structure can be stable instead of a single domain



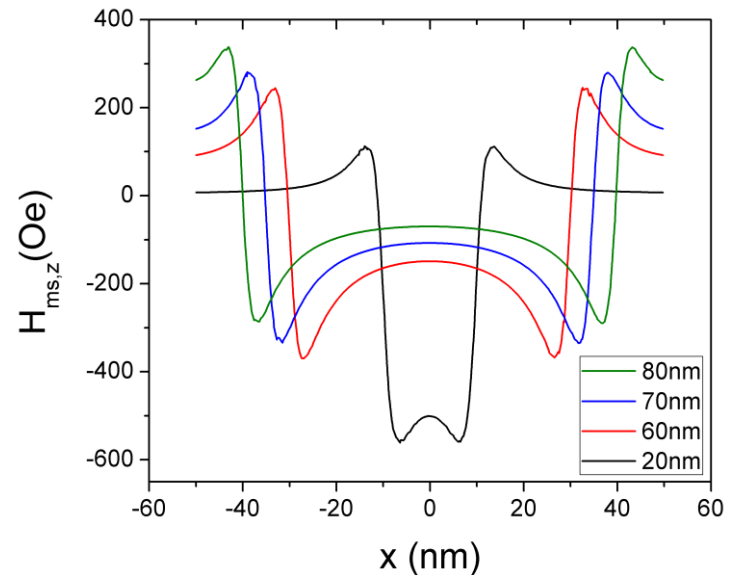
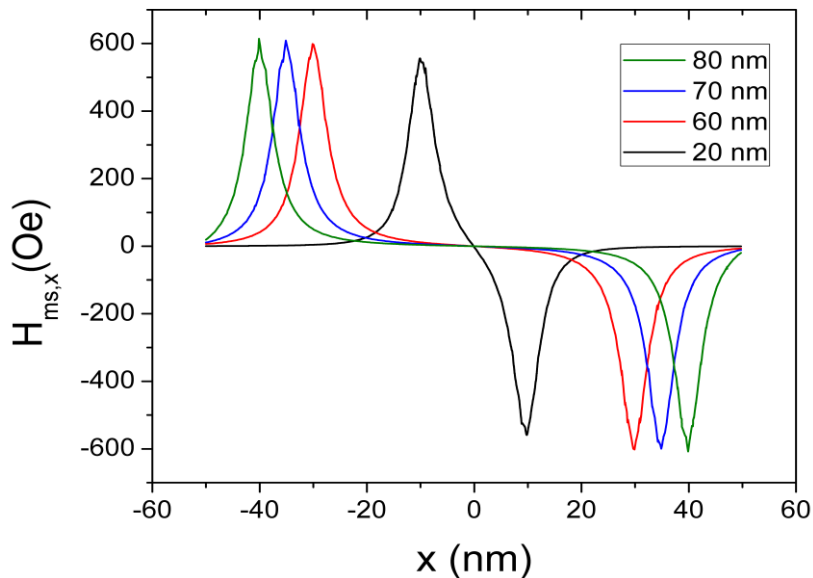
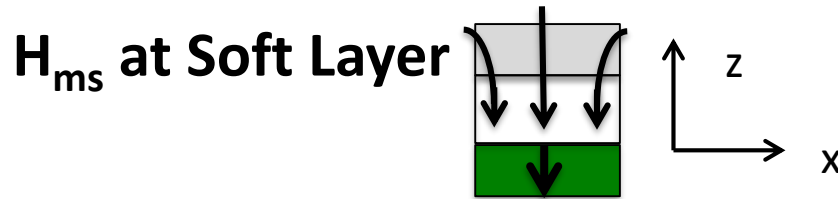
# Size Dependence of $H_c$ for a 1.5 nm Thick Disc



**Single domain** ground state below 100 nm, but doesn't reverse by simple coherent rotation due to shape  $K$

Domain wall width  $\delta \sim 100$  nm

# Stray Field from a Magnetic Disc in a Perpendicular Tunnel Junction

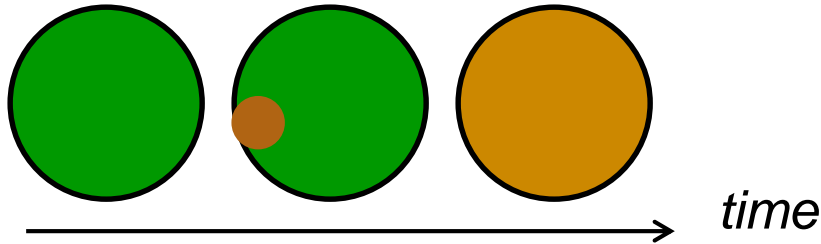


$H_{ms,x}$  large near edges but weakly size-dependent

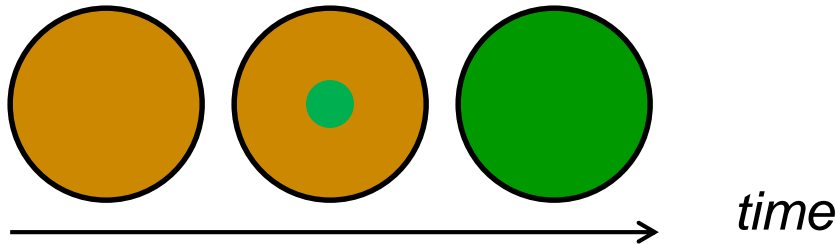
$H_{ms,z}$  size-dependent at both edges and interior

# Reversal in Magnetic Tunnel Junctions

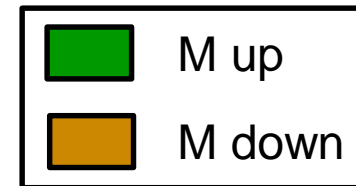
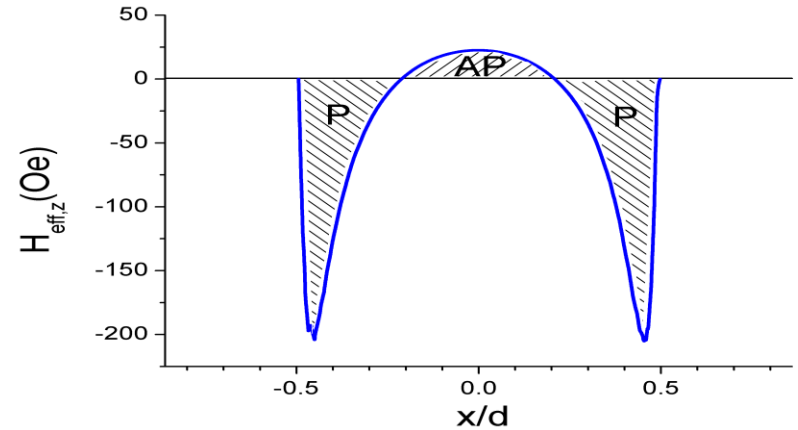
Top View of Soft Layer



AP  $\rightarrow$  P switching nucleates at edge



P  $\rightarrow$  AP switching nucleates in interior

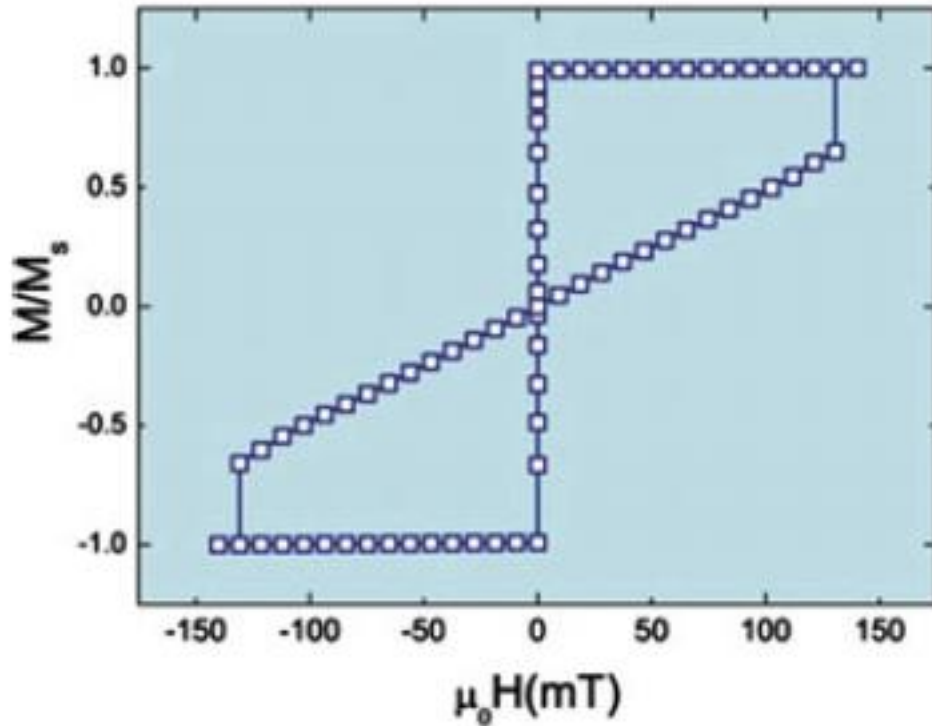


**Different energy barriers for switching “1” to “0” and “0” to “1” in perpendicular MR devices**

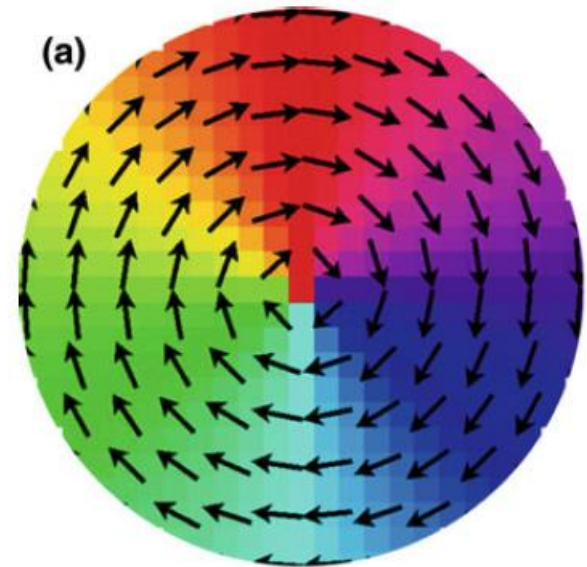


# How Can You Tell If There's a Vortex State?

Hysteresis Loop with In-Plane Field



Top View of Vortex



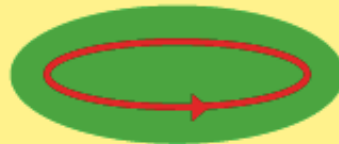
A. Guimaraes, Nanomagnetism

$H = 0$ : centered vortex

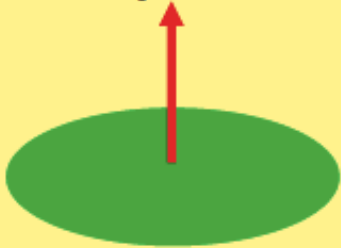
$H > 0$ : shift vortex core to edge and eliminate, leaving monodomain pattern

# Properties of magnetic vortices

- Circulation:  
 $c = +1$  (CCW)  $c = -1$  (CW)



- Polarity:  
 $p = +1$



Combining  $c$  and  $p$ :

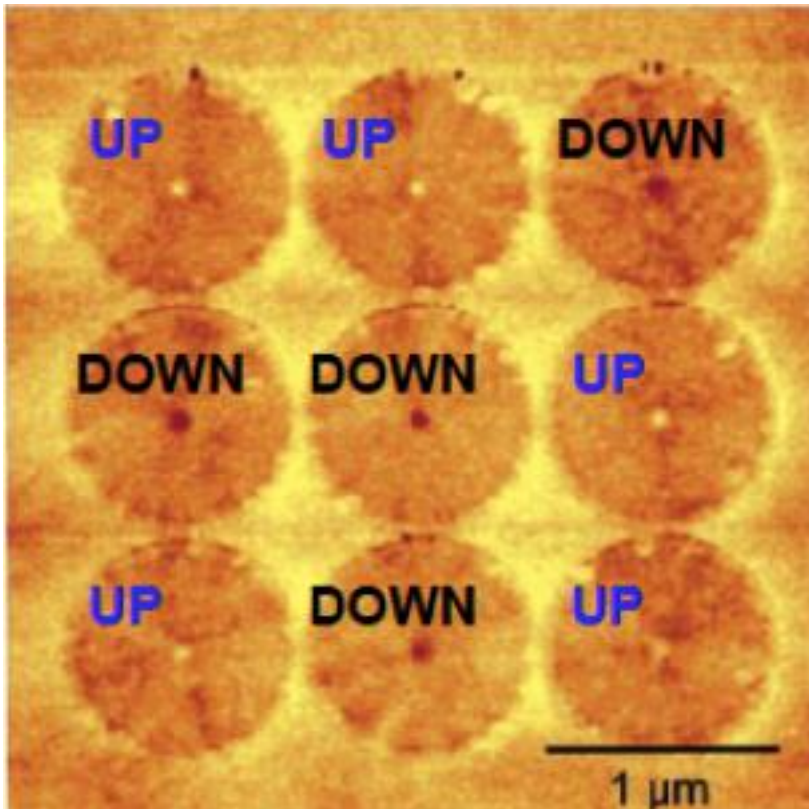
chirality or handedness:

$$cp = \pm 1$$



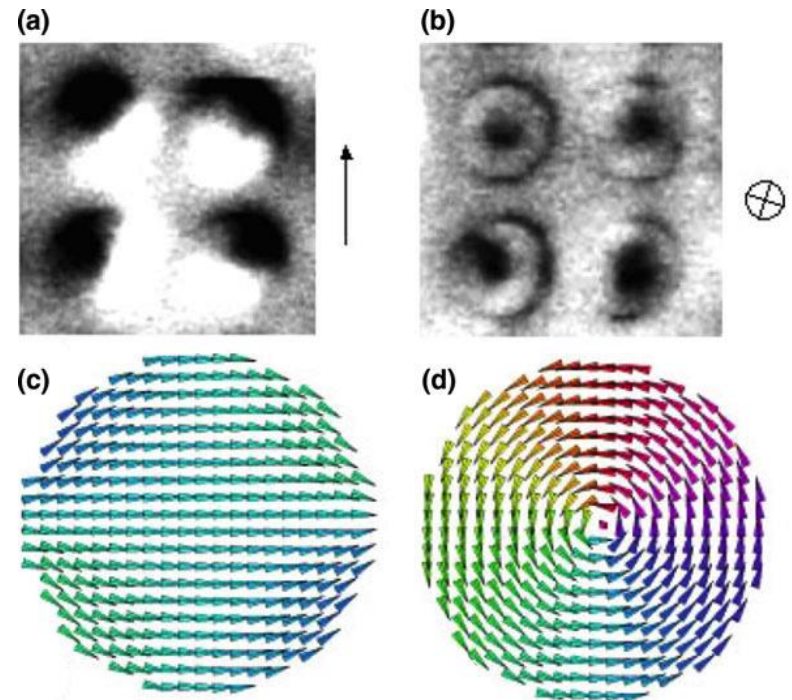
# Magnetic Force Microscopy of Vortices

Vortex cores at remanence



T. Shinjo, et al., Science **289**, 930 (2000)

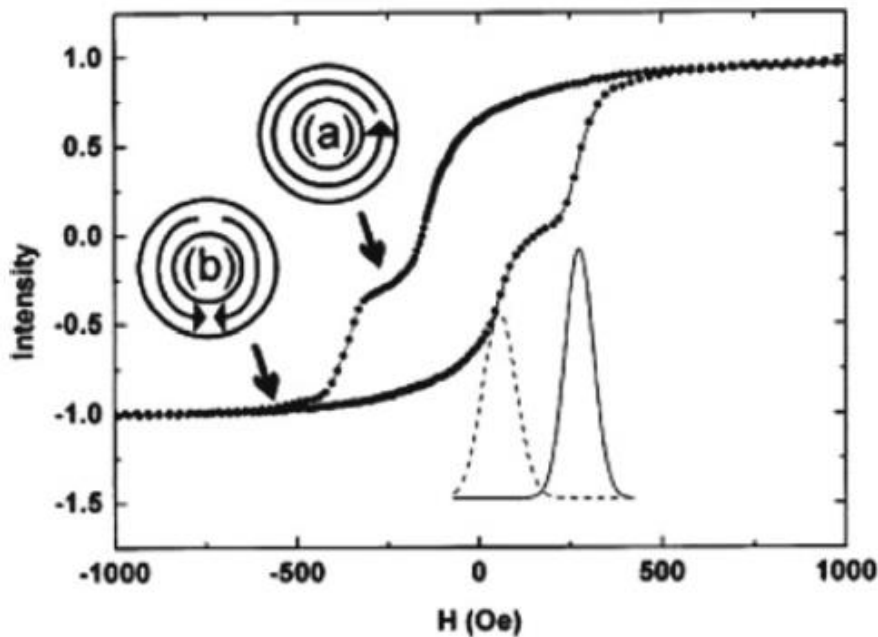
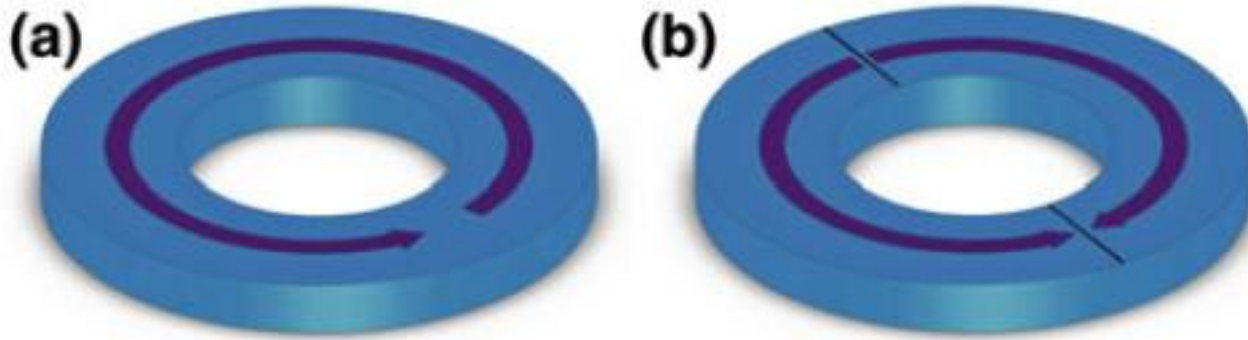
MFM Images



Single Domain

Vortex

# Magnetic Rings: Vortex versus Onion State



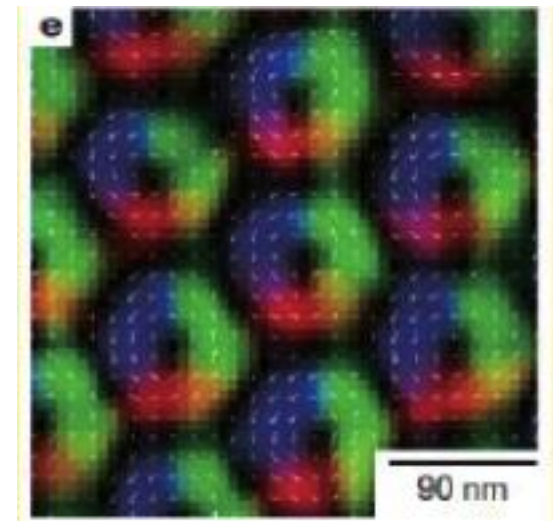
Onion state has tail-to-tail and head-to-head domain walls

# Last But Not Least: Skyrmions

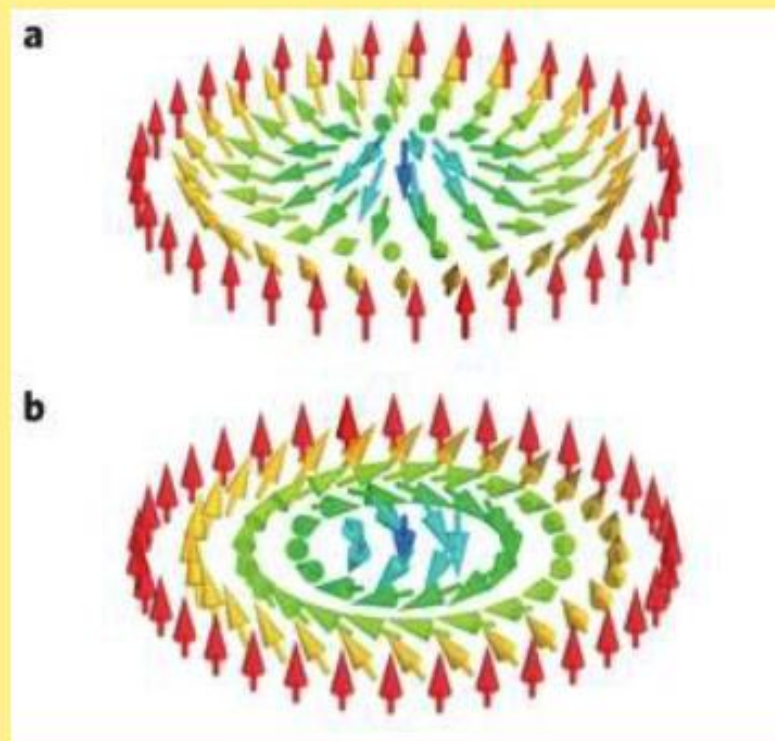
Skyrmions exist only because of Dzyaloshinskii-Moriya exchange, which requires there in no inversion symmetry

First observed at low temperature in special B20 crystals without inversion symmetry, now seen in multilayer thin films at 300 K

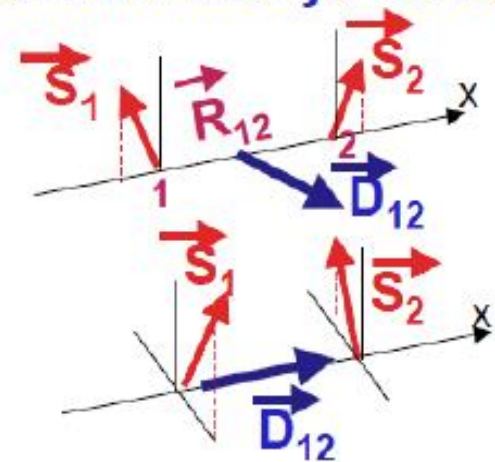
Yu (2010); Fresnel Lorentz microscopy of skyrmion lattice in  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{Si}$



# Dzyaloshinskii–Moriya interactions



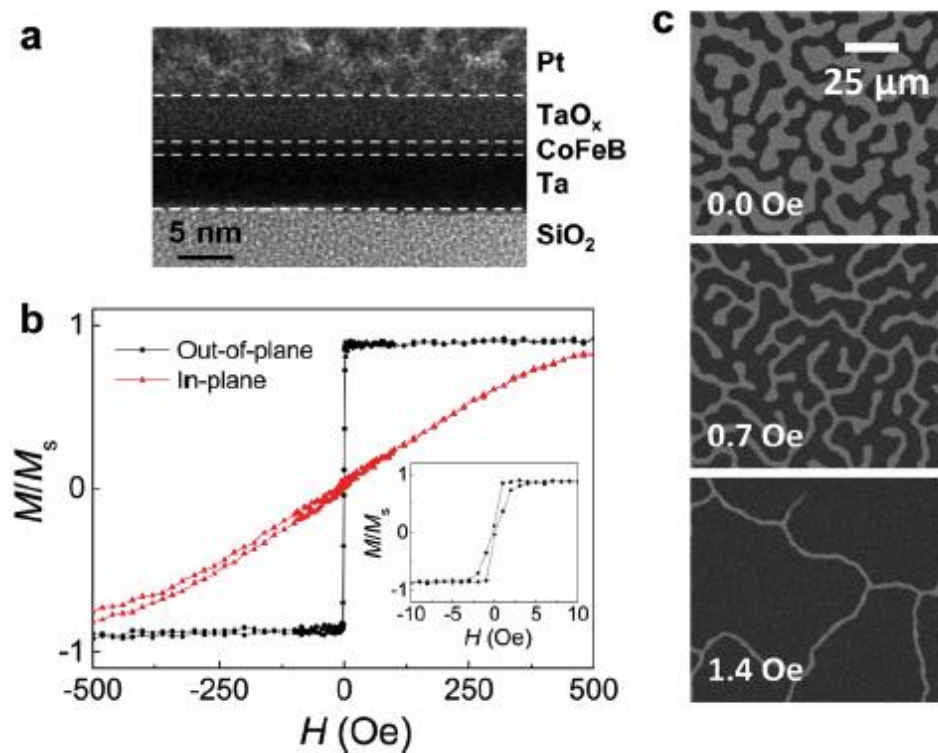
## Dzyaloshinskii-Moriya interactions



$$H_{DM} = (\vec{S}_1 \times \vec{S}_2) \cdot \vec{D}_{12}$$

# Skymionic Thin Films

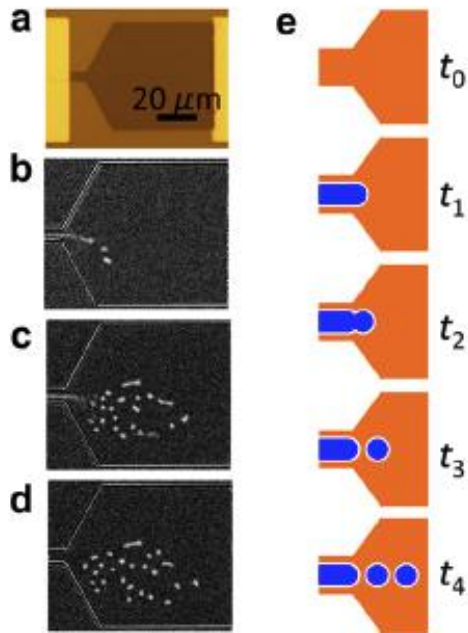
Start with a multilayer film with perpendicular K and asymmetric interfaces with strong DMI



In a continuous film, reversal occurs by the formation and growth of labyrinth domains

# How do you make Skyrmions?

Pattern the skyrmionic film stack with a narrow region leading to a wider channel



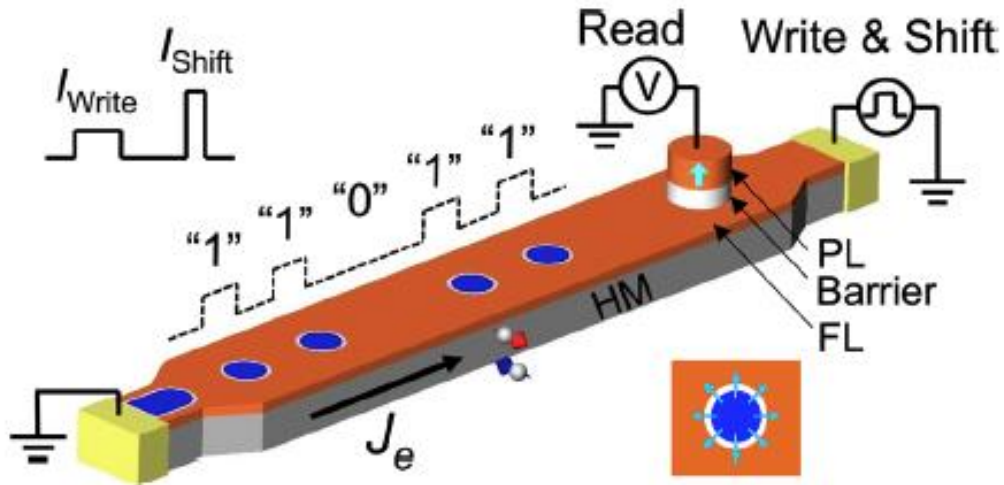
Apply a small perpendicular field sufficient to form a labyrinth domain in the narrow region but not elsewhere

Apply a DC current along the channel to apply spin transfer torque (STT) to the domain wall

In analogy to blowing bubbles due to surface tension, the current generates a stream of skyrmions



# What could you do with Skyrmions?



G. Yu, et al.,  
Nano Lett. **17**, 261 (2017)

Ideally, they are topologically protected and can be moved by a smaller current density than a domain wall, making them interesting for race track memory

Ideally they could be only a few nm in diameter (now 100s of nm)

...stay tuned!

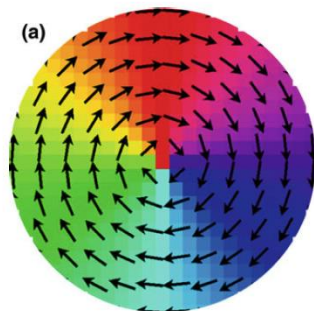
# Physics of Nanomagnetism

## Summary

The concept of a single domain is relevant to the understanding of many different kinds of nanostructures, though each has unique complexities and capabilities



Vortices and skyrmions are also important nanomagnets. There is a lot of physics yet to be understood, especially with skyrmions.



# Acknowledgments

Alberto P. Guimaraes --- For much more, see *Principles of Nanomagnetism*, 2<sup>nd</sup> edition, Springer, 2017

Bethanie Stadler – University of Minnesota

Many past and present Majetich group students and collaborators